## Lecture 8

**Chemical/Electronic Structure of Glass** 

Syllabus Topic 6. Electronic spectroscopy studies of glass structure

## Fundamentals and Applications of X-ray Photoelectron Spectroscopy (XPS) a.k.a. Electron Spectroscopy for Chemical Analysis (ESCA)



## **Review of Lecture 7**

XPS data come with Auger as by-product.

XPS of solids consists of core levels and valence band.

Intensity of core levels decreases with decreasing BE.

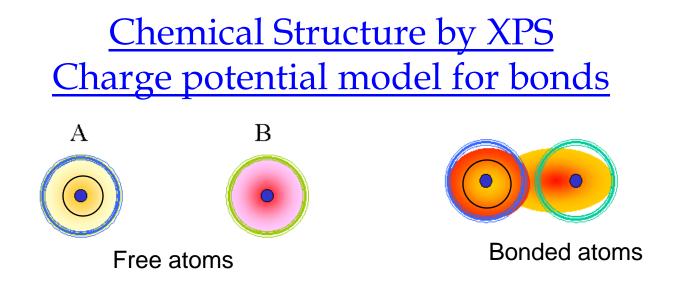
BE of a given level is unique to the particular element.

ARXPS allows depth profile by varying the angle between the detector and sample surface normal.

XPS probes <10 nm of the surface region.

The area under the peak of a core level peak is directly proportional to the concentration of that particular element.





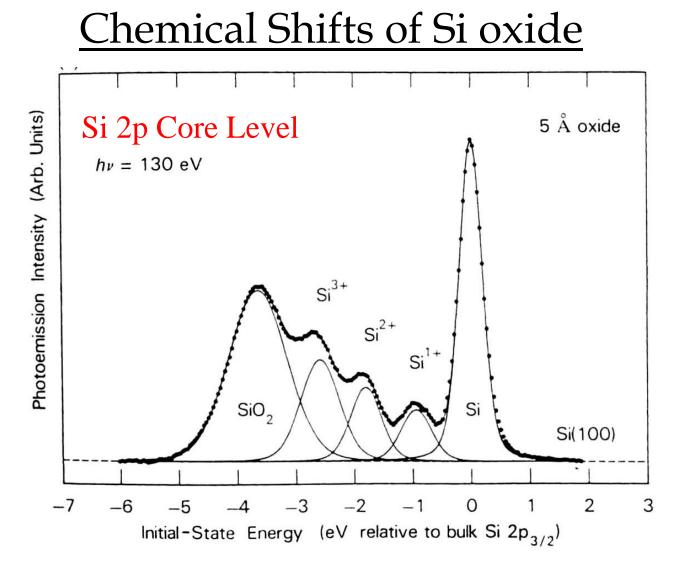
Consider valence electrons as hollow charged sphere. Neglecting relaxation effects:

$$E_{i} = E_{i0} + q_{i}/r_{v} + \Sigma_{i=j} q_{i}/r_{ij}$$
$$\Delta E_{i} = \Delta q_{i}/r_{v} + \Delta(\Sigma_{i=j} q_{i}/r_{ij})$$

 $\Delta q_i$  for valence electrons => change in energy of all inner level by  $\Delta q_i/r_v$  where  $r_v$  is the valence shell radius.

http://ligand-depot.rcsb.org/marvin/chemaxon/marvin/help/Charge.html

<sup>b</sup> Formation and Structure of Glass Feb 12, 2007.

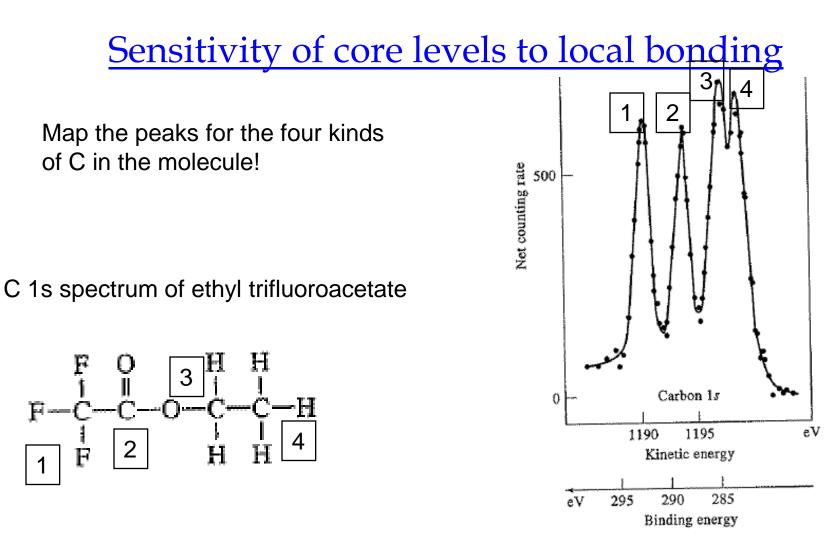


#### • Binding energy increases with increasing oxidation state of the cations.

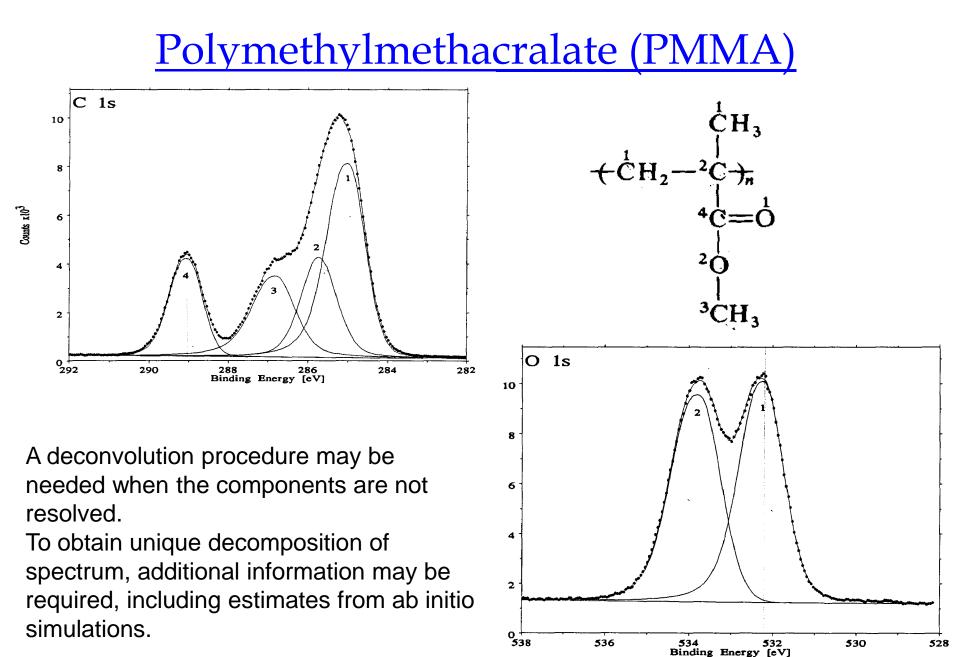
http://www.emsl.pnl.gov/new/emsl2002/tutorials/engelhard\_xps.pdf

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IMI NFG

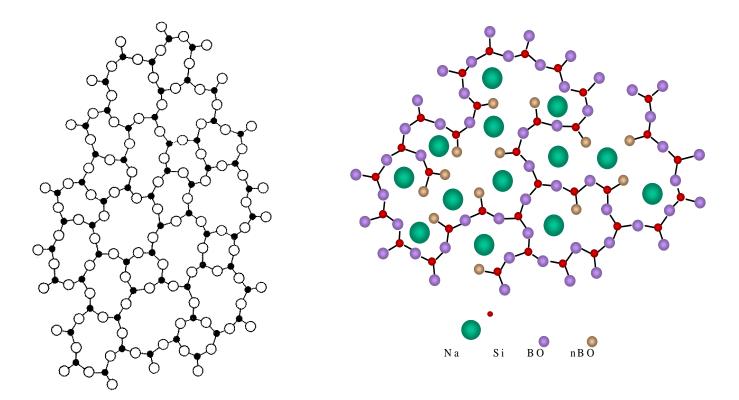


**Figure 21-5** Carbon 1s X-ray photoelectron spectrum for ethyl trifluoroacetate. (From K. Siegbahm et al., ESCA: Atomic, Molecular, and Solid-State Studies by Means of Electron Spectroscopy, p. 21. Upsala: Almquist and Wiksells, 1967. With permission.)



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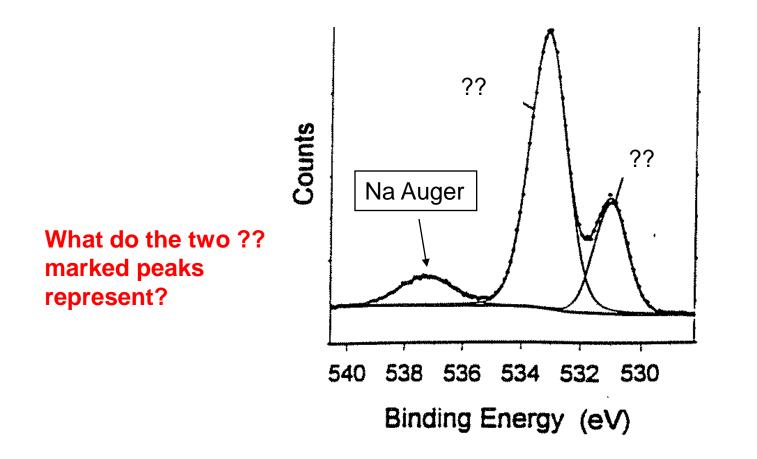
## Silica glass structure modification by alkali oxide addition





## Formation of NBO with the addition of M<sub>2</sub>O

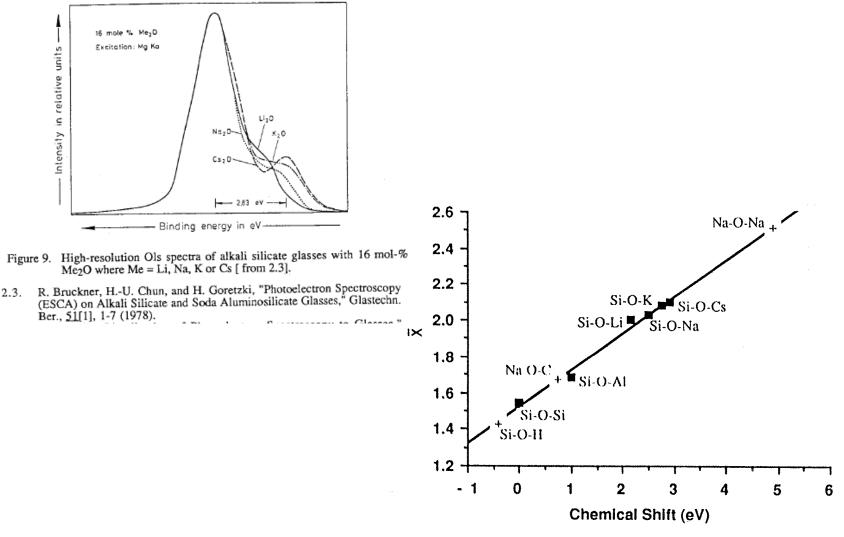
O 1s spectrum of sodium silicate glass



C. H. Hsieh et al. J. Non-cryst. Solids 168, 247-257 (1994).



# <u>O 1s Chemical shift ( $\Delta_{NBO-BO}$ ) in silicate glasses</u>



nical shift of the  $O_{1s}$  photoelectron band between non-bridging and bridging oxygens as a function of the mean electronegativity differences ( $\overline{X}$ ).

Nasu et al. JNCS 99, 140 (1988)

 $^{\diamond}$  Formation and Structure of Glass Feb 12, 2007.

### O 1s spectra of sodium silicate, borate, germanate

### and tellurite glass series

(a).  $xNa_2O-(1-x)SiO_2$ (b).  $xNa_2O-(1-x)B_2O_3$ (c).  $xNa_2O-(1-x)GeO_2$ (d).  $xNa_2O-(1-x)TeO_2$ 

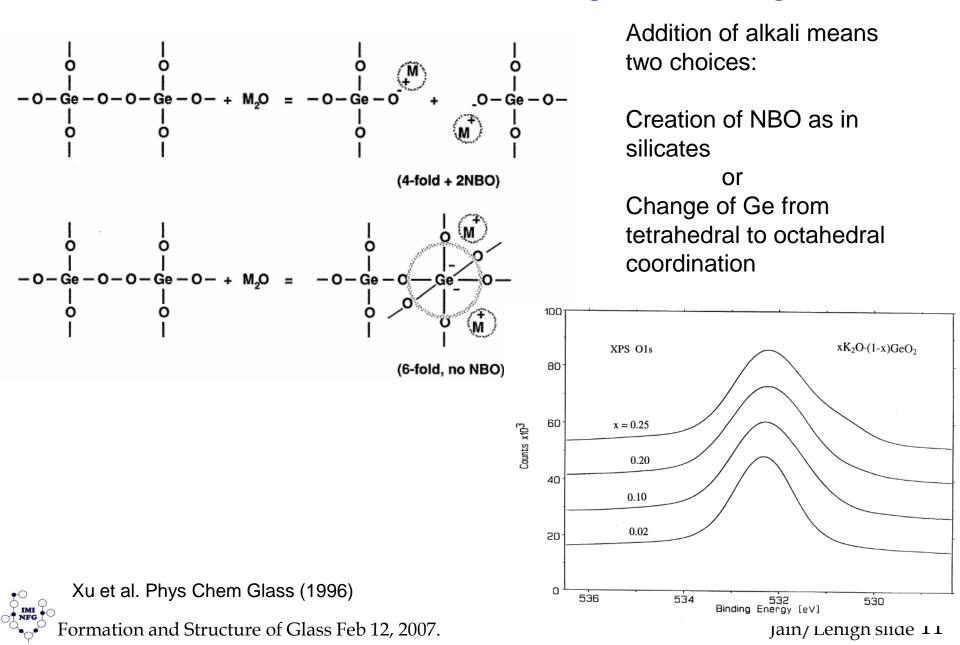
What question was raised earlier in the course with regard to the difference in the structure of alkali silicates and germanates?

(a) (b) x=0.40 units units x=0.50 0.35 0.40 0.30 Intensity / arb. arb. 0.25 0.33 ntensity 0.20 0.30 0.15 0.10 0.20 0.05 0 0 535 540 530 525 520 540 535 530 525 520 Binding energy / eV Binding energy / eV (c) (d) x=0.35 ntensity / arb. units units x=0.30 0.30 arb. 0.25 0.25 Intensity 0.20 0.15 0.15 0.07 0.10 0 0 535 530 525 Binding energy / eV 540 535 530 520 540 525 520 Binding energy / eV

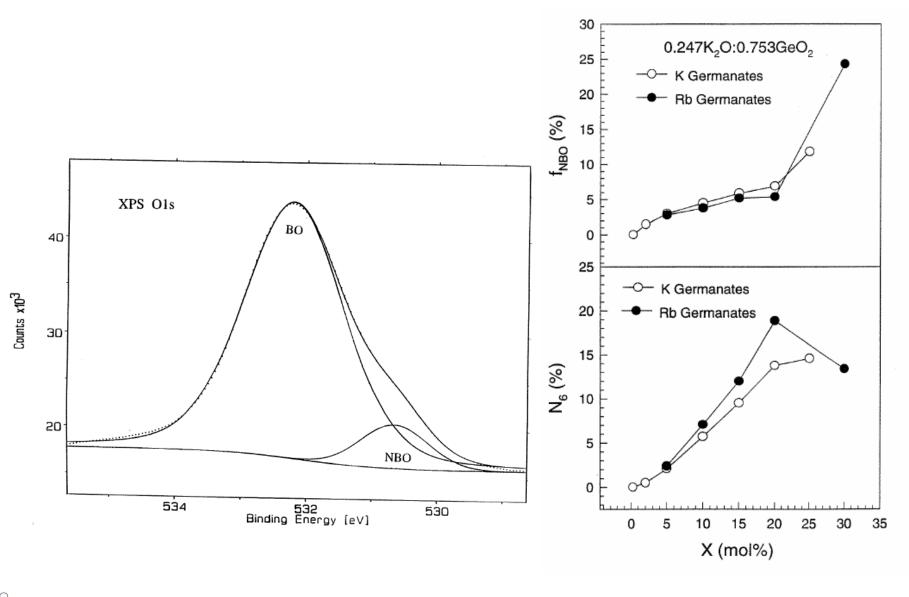
Nanba and Miura

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### Network modification in alkali germanate glasses



### NBOs vs Ge-octahedra?



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# <u>O-1s BE for SiO<sub>2</sub></u>

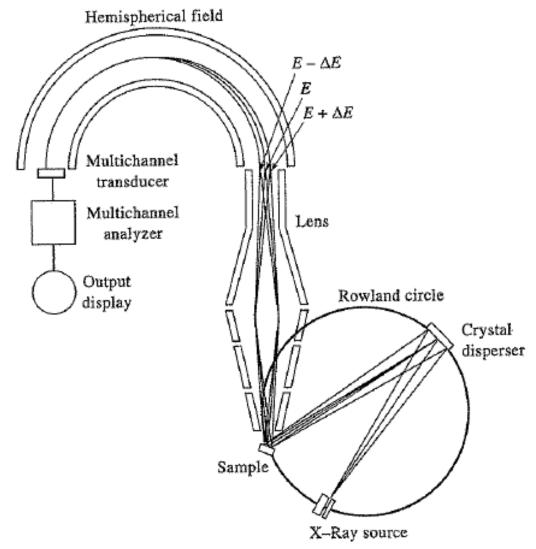
533.0 Polycrystalline powder of stishovite, natural mineral from meteor crater Arizona, the densest modification of SiO2

- 533.2 Alpha-quartz
- 532.9 pelletized, composition determined by XPS is SiO2.08
- 533.8 Thermal oxide SiO2.1.
- 532.7 Thermaly grown SiO2
- 532.7 Fused quartz.
- 532.8 alpha phase, insulator, polycrystalline
- 533.2 Quartz (rock crystal).
- 532.0 Polycrystalline powder of stishovite from meteor crater, Arizona

#### NIST X-ray Photoelectron Spectroscopy Database: http://srdata.nist.gov/xps/



### XPS instrument schematic

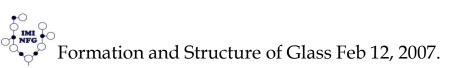




### **Complications of XPS on glass**

Charging of surface Error in BE Error in composition

Damage by sputtering (if used in depth analysis) Alteration of structure



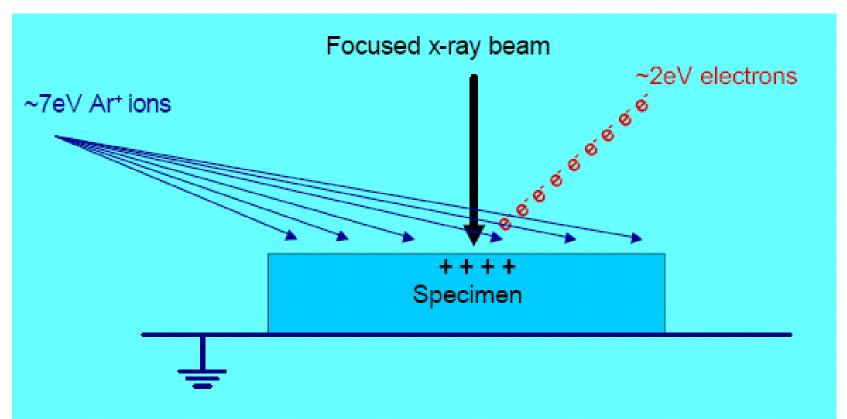
# <u>Charging of insulating sample</u>

Net build up of charge on the surface when e loss is not compensated by inward flow => Surface at unknown +V => All peaks shifted/broadened by ~ the same amount.

Problem mostly corrected by flooding with low energy e.

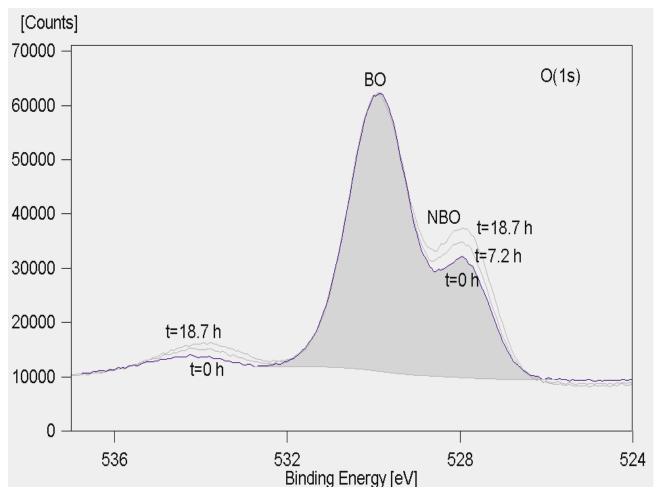
### For precise b.e. values, need a reference:

- 1. Adventitious C-1s (may not be present on pristine surface)
- 2. Thin overlayer of Au
- 3. Internal reference e.g. Si-2p in silicate glasses.
- 4. Use Auger parameter



- Low-energy electrons from a cold cathode flood gun alleviates positive charging
- Low-energy source of positive ions alleviates the surrounding negative charge

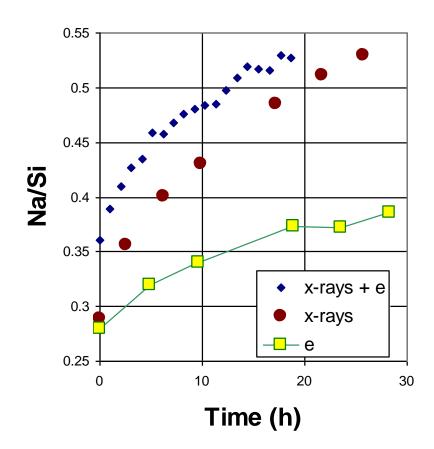
# <u>f<sub>NBO</sub> vs. time of XPS experiment</u>



Soda-Lime-Silicate Glass 13.30Na<sub>2</sub>O 11.62CaO 73.86SiO<sub>2</sub> (mole %)

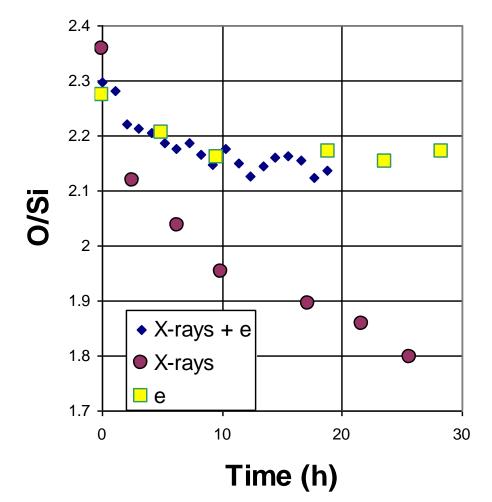


## Na conc. vs time



X-irradiation is the primary cause for Na accumulation. Once liberated, Na can migrate to the surface very quickly, if e-field is present.

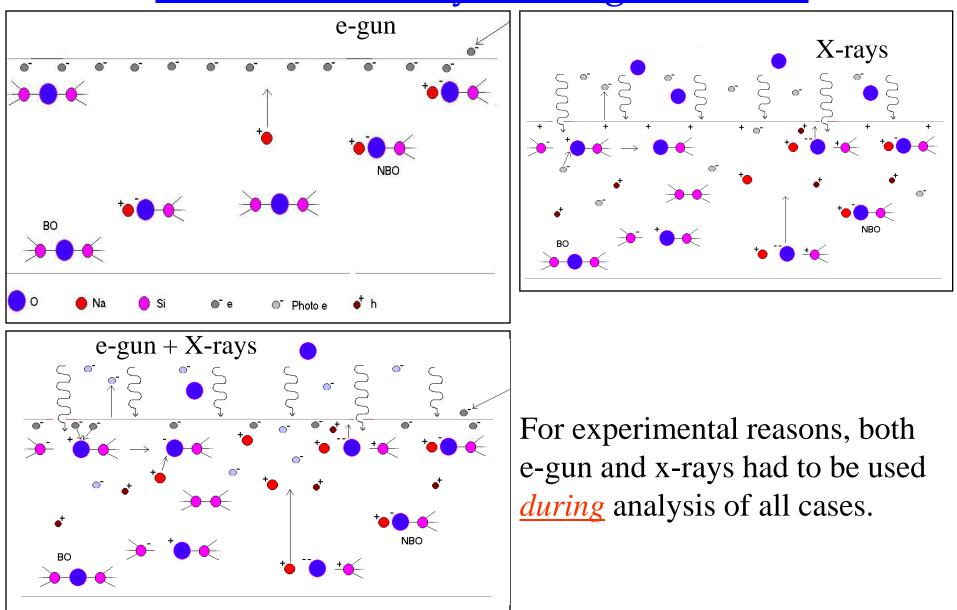
## Total oxygen vs time



The loss of oxygen is mostly determined by x-ray dose; electrons retard it.

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## Contrast of x-ray and e-gun effects



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## Summary: XPS capabilities

- Elements detected from Li to U.
- Nondestructive (x-ray beam damage in certain materials?)
- Quantitative.
  - Chemical bonding (e-density) analysis.
- Surface sensitivity from 5 to 75 angstroms.
- Conducting and insulating materials.
- Detection limits that range form 0.01 to 0.5 atom percent.
- Spatial resolution for surface mapping from >10  $\mu$ m.
- Depth profiling (non-destructive as well as destructive).