

# Optical and Photonic Glasses

## **Lecture 6: Structures of Glass II – Glass Families**

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## Designations for **modified silicate glasses**

Example of alkaline-earth silicates ( $\text{MO} \cdot \text{SiO}_2$ , in mol%):

	<u>example</u>	<u>designation</u>	<u>net. dimens.</u>	<u># BO/tetr.</u>	<u>symbol</u>
0 %	$\text{MO} \rightarrow \text{SiO}_2$	(silica)	3-D	(4 BO/tetr.)	$Q_4$
33 %	$\text{MO} \rightarrow \text{CaO} \cdot 2 \text{SiO}_2$	(disilicate)	2-D	(3 BO/tetr.)	$Q_3$
50 %	$\text{MO} \rightarrow \text{CaO} \cdot \text{SiO}_2$	(metasilicate)	1-D	(2 BO/tetr.)	$Q_2$
60 %	$\text{MO} \rightarrow 3 \text{CaO} \cdot 2 \text{SiO}_2$	(pyrosilicate)	dimers	(1 BO/tetr.)	$Q_1$
66 %	$\text{MO} \rightarrow 2 \text{CaO} \cdot \text{SiO}_2$	(orthosilicate)	monomers	(0 BO/tetr.)	$Q_0$

## Modified silicate glasses

### A) Fraction of NBO atoms ( $f_{\text{NBO}}$ )

molar basis:  $20 \text{ Na}_2\text{O} \cdot 80 \text{ SiO}_2 = \text{Na}_2\text{Si}_4\text{O}_9$

$1 \text{ Na}^+ \Rightarrow 1 \text{ NBO}^-$       ( $1 \text{ Ca}^{2+} \Rightarrow 2 \text{ NBO}^-$ )

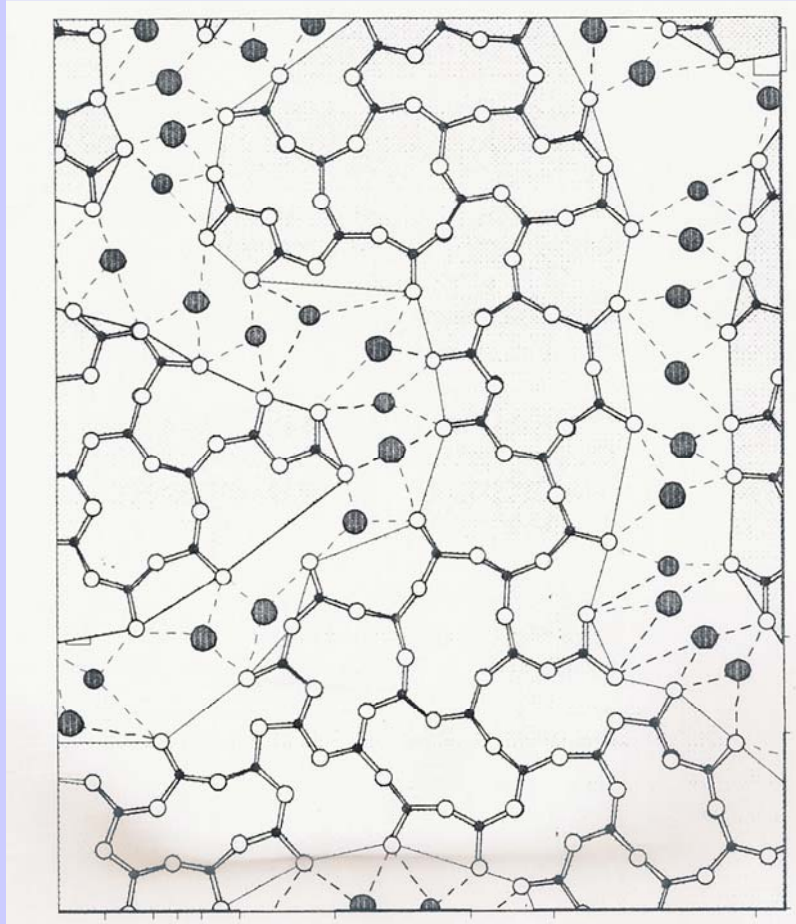
$$f_{\text{NBO}} = 2/9 = 22\% \quad (0.22)$$

### B) Average number, $y$ , of corners shared, or **BO**, per tetrahedron (for particular compositions, this coincides with the subscript $n$ in $Q_n$ symbols):

$$y = 6 - (200 / \text{mol}\% \text{ SiO}_2) \quad (\text{only for tetrahedral silicates})$$

Example: in a *disilicate* glass, the basic structural unit is a  $Q_3$  “tetrahedron” (strictly speaking, a trigonal pyramid, where the Si-O<sup>-</sup> bond is shorter than in Si-O-Si bonds), possessing (3 BO + 1 NBO) species and a total of 2.5 (net) oxygen atoms. Hence its chemical formula:





This is a 2-D representation of the *modified random network* model of G.N. Greaves et al. (The structure of Non-Crystalline Materials, Taylor & Francis, 1983) for a modified silicate glass

- Si
- Na
- O O

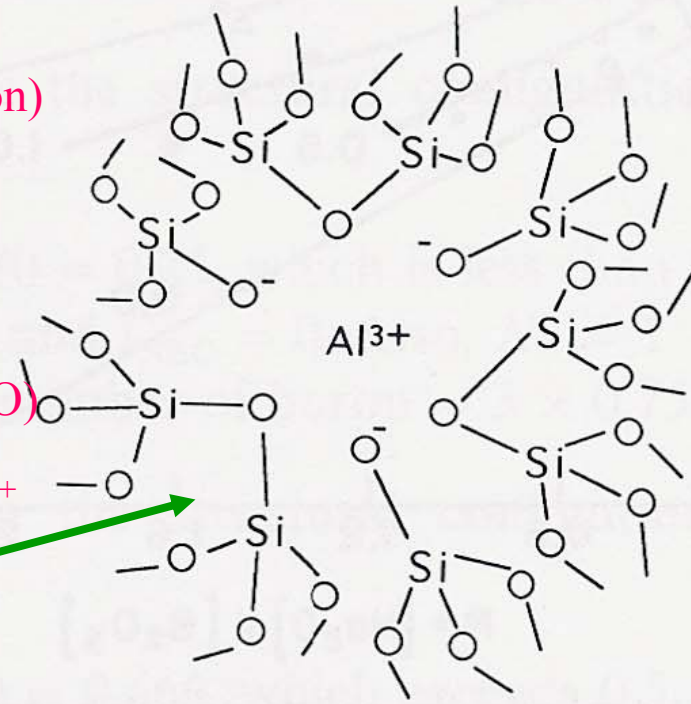
In this model, there are **channels** along which the **modifier ions** ( $\text{Na}^+$ , in this case) are preferentially located, corresponding to a certain extent of *phase separation* in the modified glass.

# Alkali aluminosilicate glasses: 4- and 6-fold coordinated Al

$\text{Al}_2\text{O}_3 / \text{M}_2\text{O} < 1 \Rightarrow \text{CN}_{\text{Al}} = 4$   
(excess negative charge of  $[\text{AlO}_4]$  group satisfied by  $\text{M}^+$  ion)

$\text{Al}_2\text{O}_3 / \text{M}_2\text{O} = 1 \Rightarrow \text{CN}_{\text{Al}} = 4$   
(network **former**) and no NBO present (adding one  $\text{Al}^{3+}$  to an alkali silicate removes one NBO)

$\text{Al}_2\text{O}_3 / \text{M}_2\text{O} > 1 \Rightarrow$  excess  $\text{Al}^{3+}$  are **modifiers**, with  $\text{CN}_{\text{Al}} = 6$

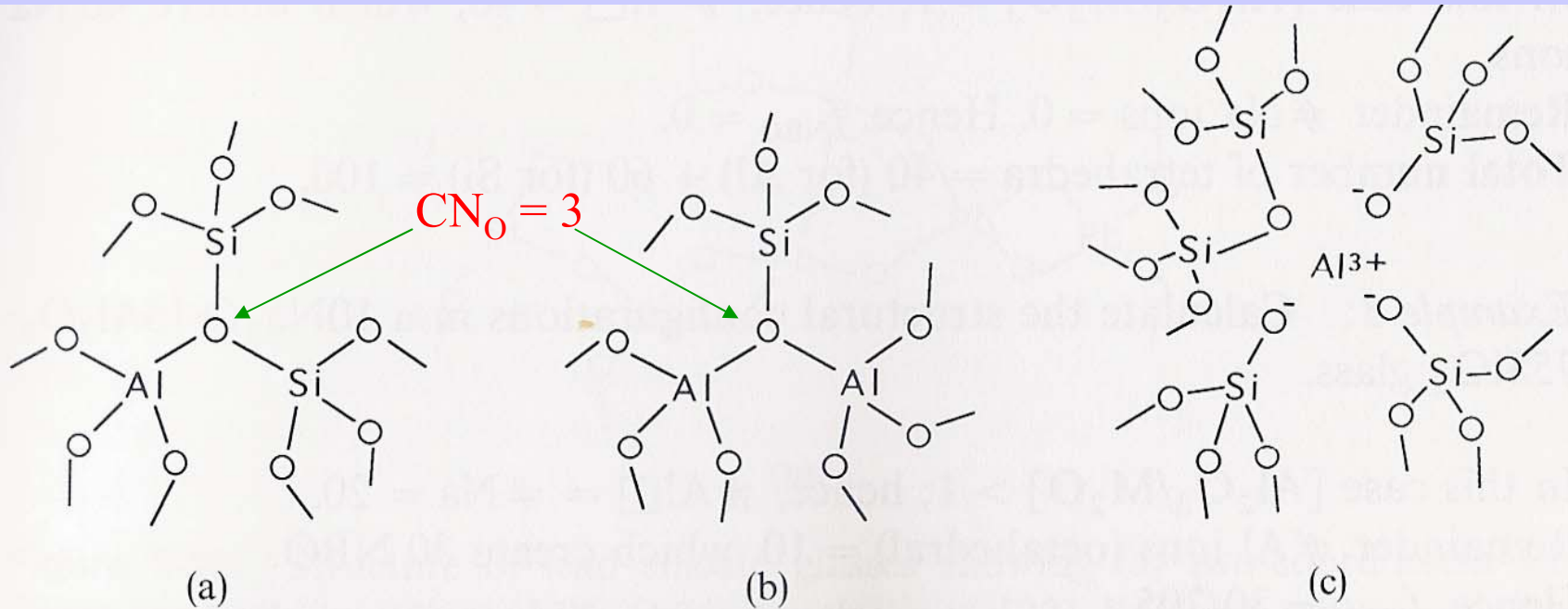


(Presumably, a modifying  $\text{Al}^{3+}$  ion, with octahedral coordination, will have its charge compensated by 3 singly charged NBO ions).

Suggested structure of an alkali aluminosilicate glass with  $\text{Al}^{3+}$  as a network modifier showing octahedrally coordinated  $\text{Al}^{3+}$ .

(Adapted from: *Fundamentals of inorganic glasses*, A.K. Varshneya, Academic Press, 1994)

# Alkali aluminosilicate glasses: 4-fold coordinated Al even for $\text{Al}_2\text{O}_3 / \text{M}_2\text{O} > 1$

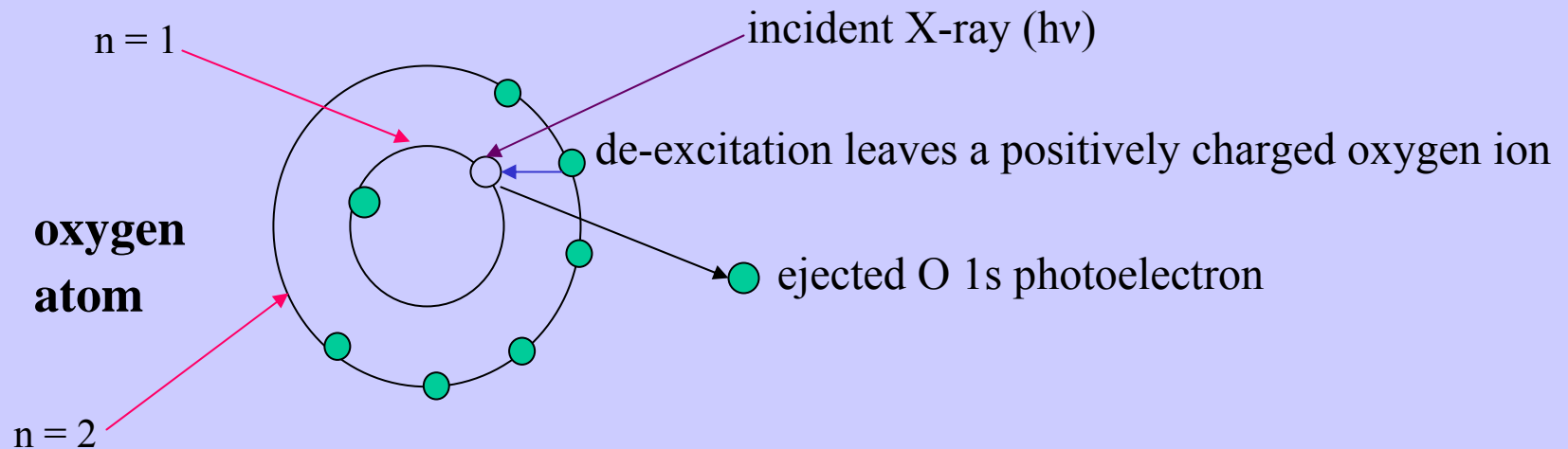


Tetrahedrally coordinated  $\text{Al}^{3+}$  in tricluster arrangements (a) and (b) according to Lacy. Note that (c) is equivalent to (a).  
(E.D. Lacy, 1963)

(Adapted from: *Fundamentals of inorganic glasses*, A.K. Varshneya, Academic Press, 1994)

# X-ray photoemission (or photoelectron) spectroscopy (XPS)

XPS is one of the best available experimental tools for the identification and quantitative determination of NBO species in oxide glasses.

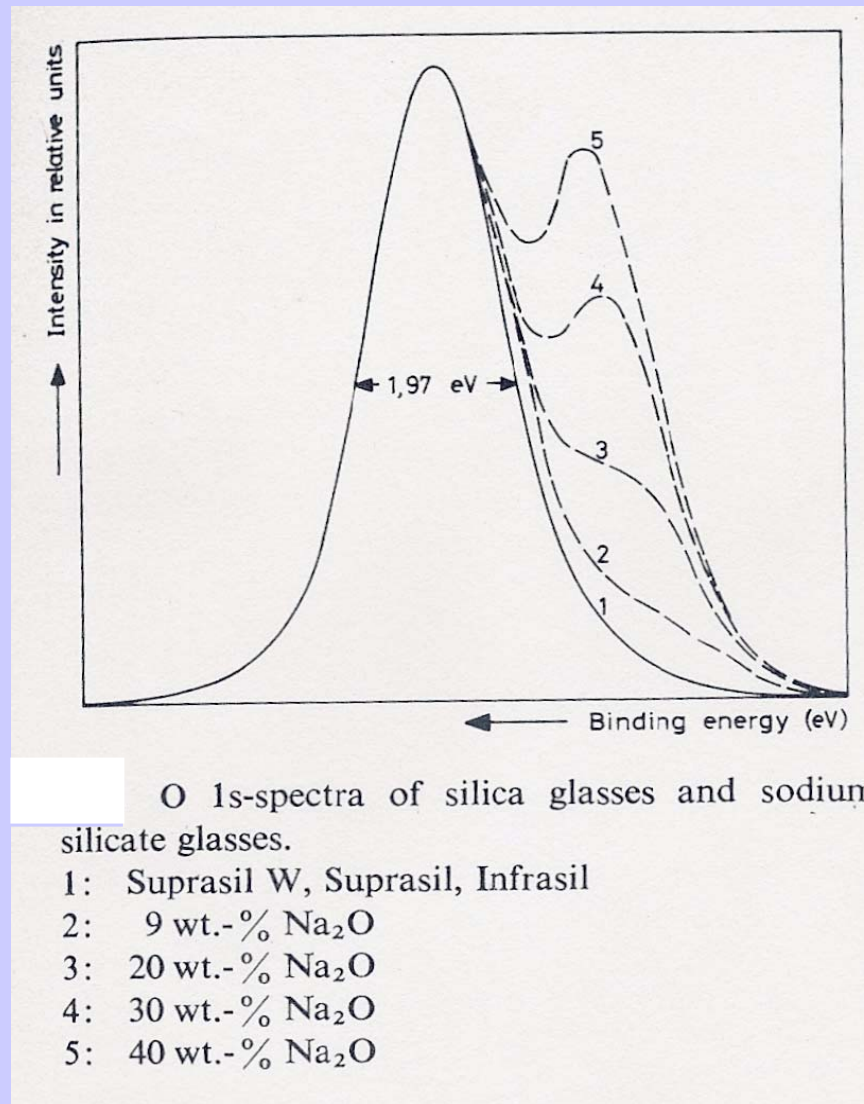


Photoelectron:  $BE = h\nu - KE$  (e.g.,  $O_{1s}$ :  $BE \sim 532 \text{ eV}$ )

De-excitation energy: (1) X-ray photon (XRF); (2) secondary Auger electron (KLL), leaving a doubly positively charged oxygen ion.

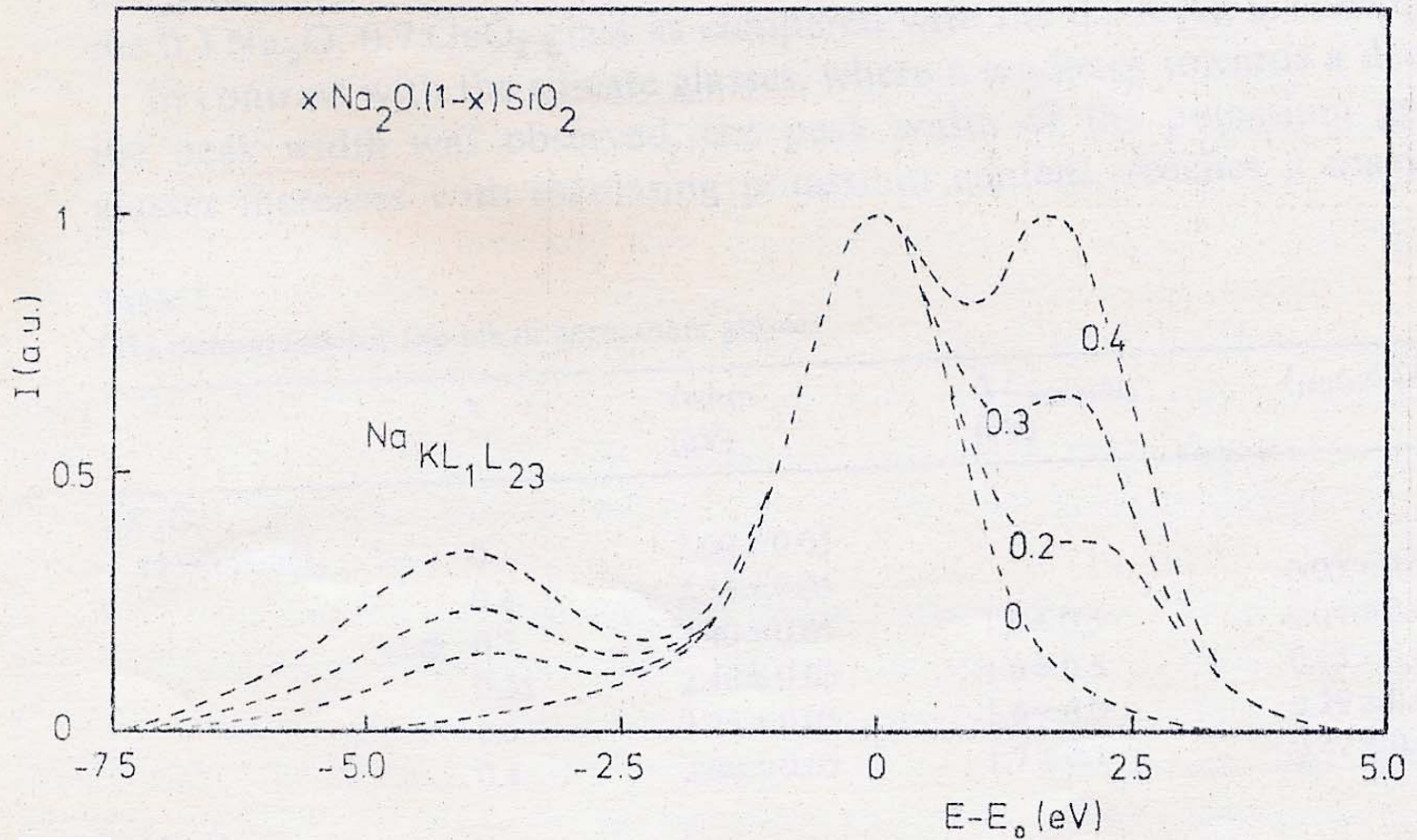


High resolution O1s XPS spectra of silica and sodium silicate glasses. (The areas under the low and high BE peaks, separated by  $\sim 2$  eV, are proportional to the NBO/BO atomic ratio).



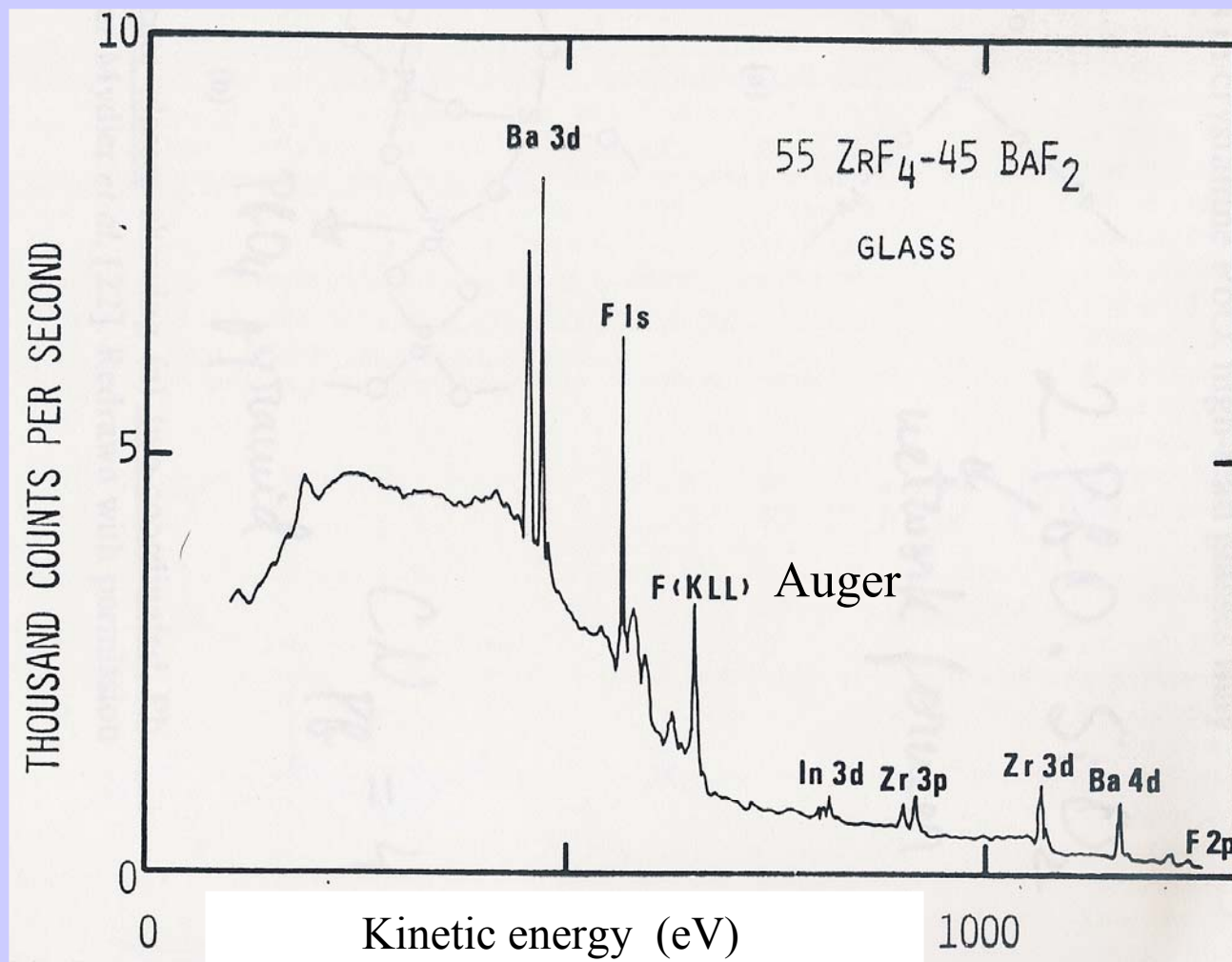
(Adapted from: R. Bruckner et al., Jap. J. Appl. Phys., 1978)





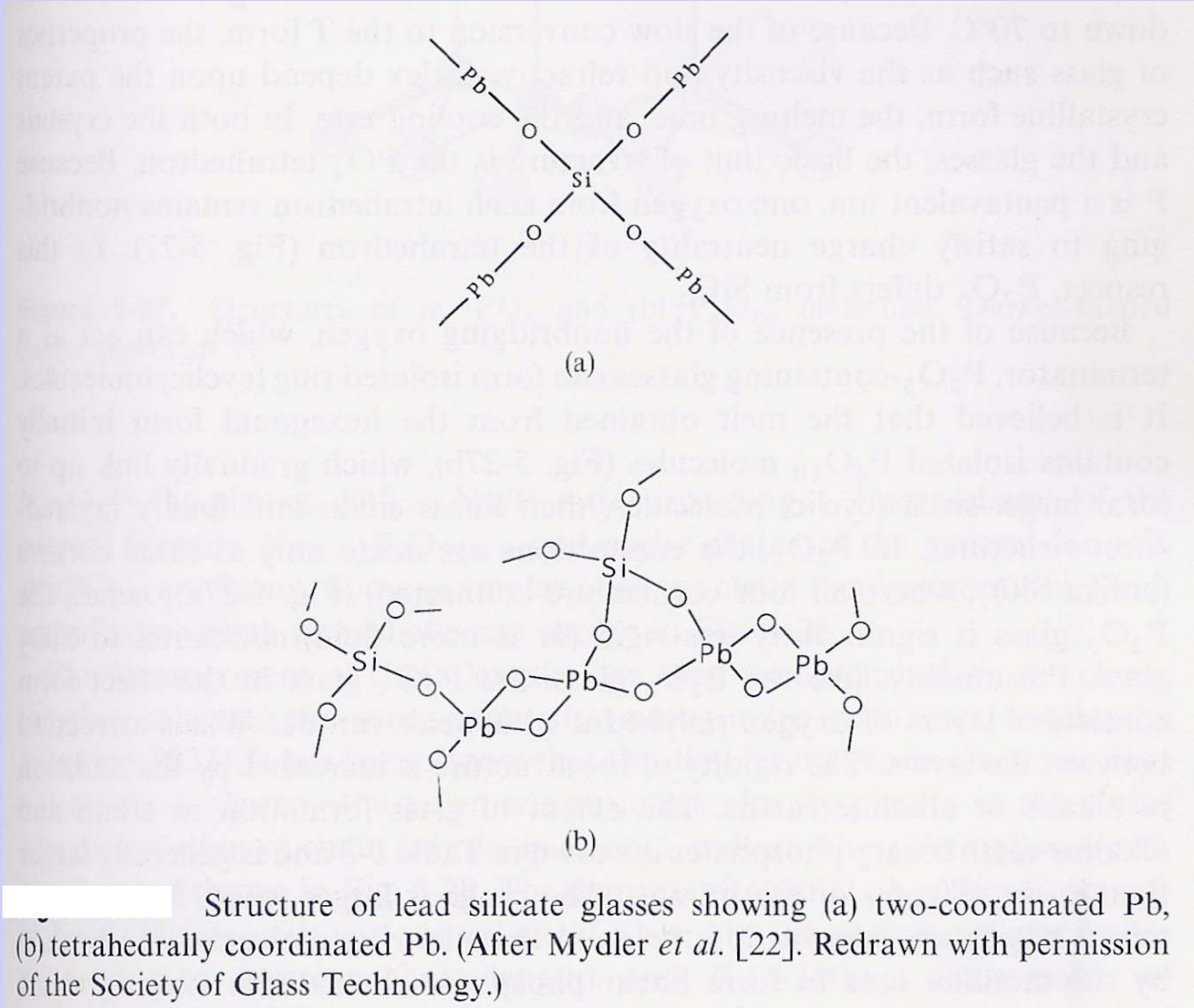
O1s spectra of the sodium silicate glasses (kinetic energy scale relative to the energy of the BO's).

(Adapted from: B.M.J. Smets and T.P.A. Lommen, J. Non-Cryst. Solids, 1981)



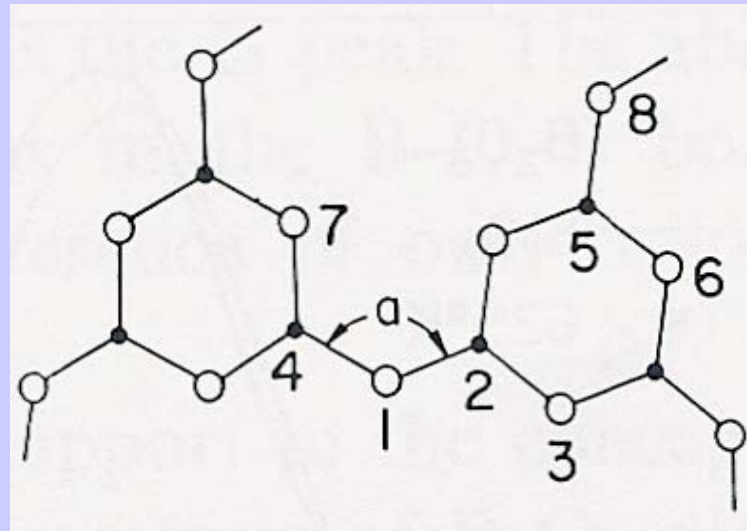
XPS may be applied to non-oxide glasses as well. E.g., ZrF<sub>4</sub>-based glasses have *bridging* and *non-bridging* fluorine ions, but with a smaller difference in BE.

Lead silicate and lead borate glasses may contain more than 90 weight% PbO. In such glasses, it is believed that  $Pb^{2+}$  cations assume a network-forming role, either coordinated by two oxygen ions (fig. (a)) or by four, forming  $[PbO_4]$  pyramids (fig. (b)).



(Adapted from: *Fundamentals of inorganic glasses*, A.K. Varshneya, Academic Press, 1994)

The structure of pure  $\text{B}_2\text{O}_3$  glass, as evidenced from detailed NMR and Raman spectroscopy studies and confirmed by XRD, appears to have the so-called *boroxol* ring as the fundamental structural unit, formed by three B atoms, each surrounded by three nearest neighbor BO atoms.

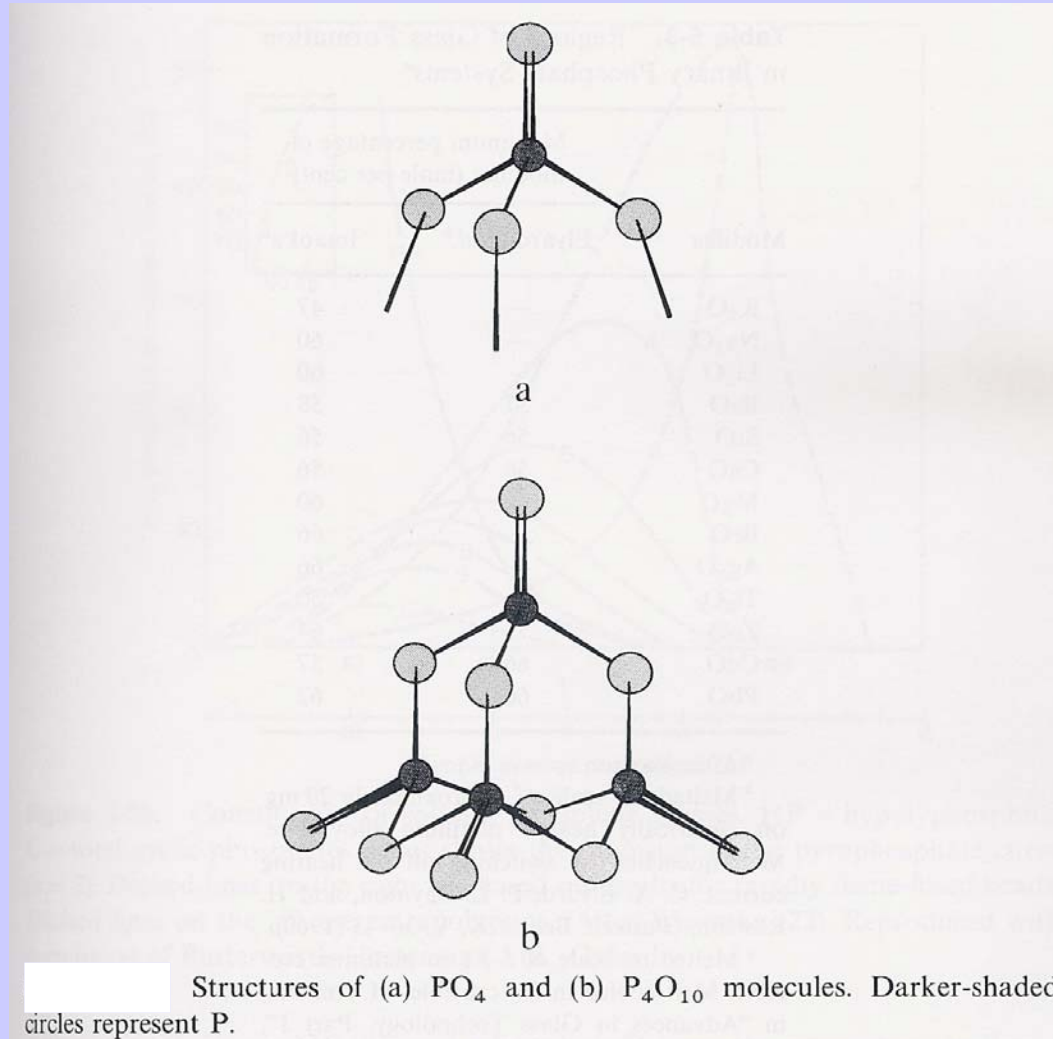


Configuration of the boroxol ring.

(Adapted from: *Fundamentals of inorganic glasses*, A.K. Varshneya, Academic Press, 1994)

As a modifier such as  $\text{Na}_2\text{O}$  is progressively added to  $\text{B}_2\text{O}_3$ , an increasing number of B atoms go into tetrahedral rather than triangular coordination. This has significant consequences, in particular a clear minimum in the thermal expansion coefficient vs.  $[\text{Na}_2\text{O}]$  curve, usually designated as the *boron oxide anomaly*.

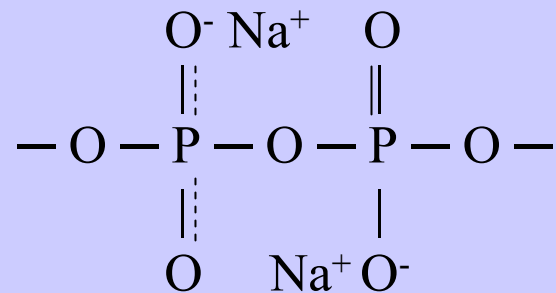
$P_2O_5$  is a glass-forming compound. The coordination of P in pure  **$P_2O_5$  glass** is believed to be similar to that found in  $PO_4$  and  $P_4O_{10}$  molecules (figs. (a) and (b)), consisting of three single bonds and a double  $P=O$  bond. This has a special type of NBO, corresponding to a terminal bond shorter than a normal  $P-O^-$  terminal bond.



(Adapted from: *Fundamentals of inorganic glasses*, A.K. Varshneya, Academic Press, 1994)



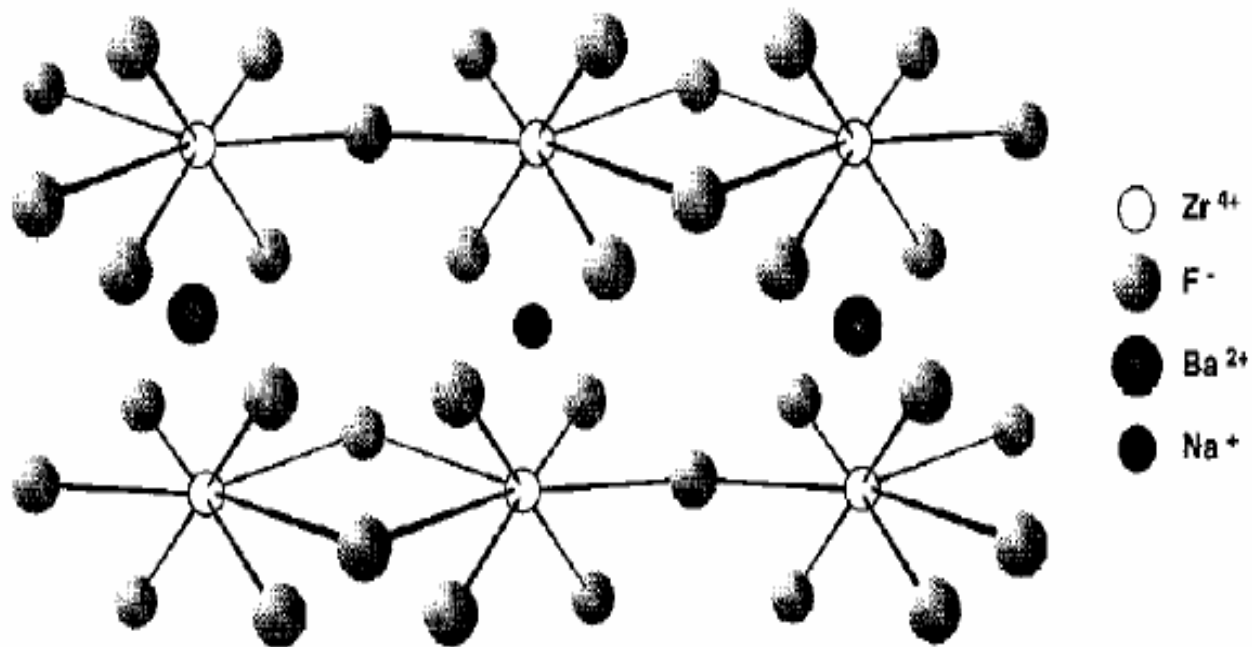
One well known type of modified phosphate glass is the *metaphosphate* glass composition, where the basic structural units are chains. In the case of sodium metaphosphate ( $\text{NaPO}_3$ ) glass, the structural unit may be represented as follows:



The out-of-chain P-O bonds have a character in-between a single and a double bond. Alternatively, this situation might be depicted by means of a regular single P-O<sup>-</sup> bond and double P=O bond on opposite sides of each P atom. The figure includes examples of both descriptions.



**Halide** glasses such as the  $\text{ZrF}_4$ -based ones are more ionic than oxide glasses and the network-forming cation,  $\text{Zr}^{4+}$ , has higher coordinations than those predicted by Zachariasen's rules ( $\text{CN}_{\text{Zr}}$  has been found to be between  $\sim 6 - 8$ ).



Chain-like skeleton in the structure of a  $\text{ZBLAN}_{66}$  glass (57.0  $\text{ZrF}_4$ -28.1  $\text{BaF}_2$ -3.3  $\text{LaF}_3$ -5.0  $\text{AlF}_3$ -6.6  $\text{NaF}$ , in mol%).

(Adapted from: M.C. Goncalves and R.M. Almeida, Mat. Res. Bull. 31 (1996) 573)