



# Optical hole burning studies in europium doped glasses\*

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- Elizabeth Schoolfield



# Outline

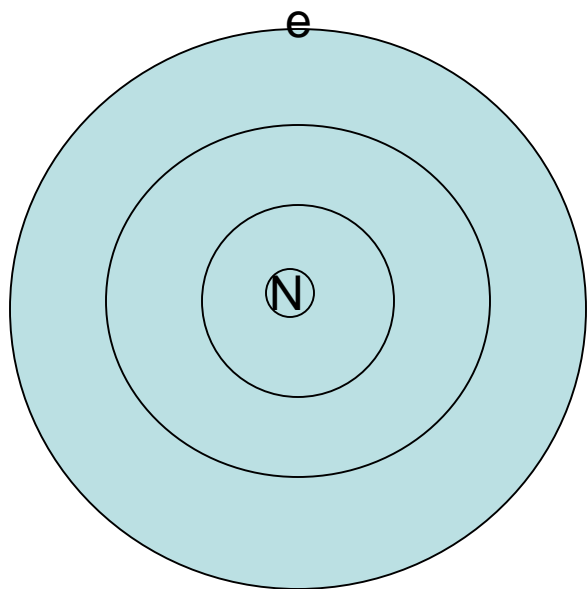


- Introduction
- Spectral Hole Burning
- Application
- Glass preparation
- Experimental Setup
- Optical Characterization and ESR
- Results
- Conclusions

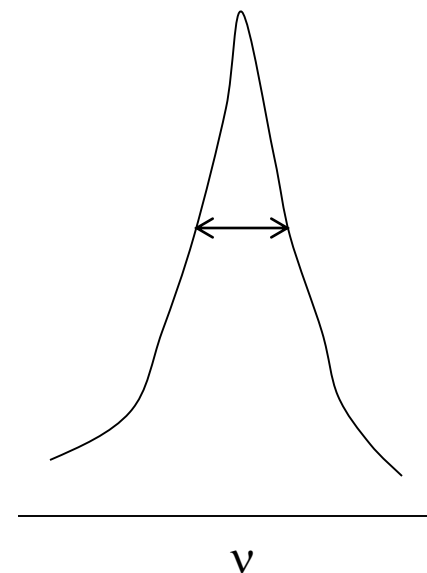
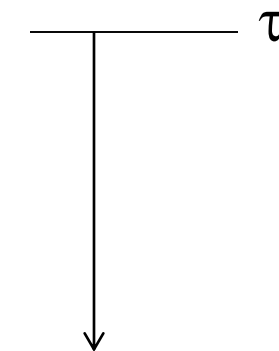
# Ionic energy levels



- Electron orbits



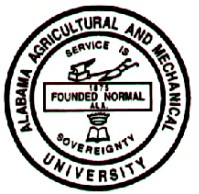
- Energy levels



# Units: What is the resolution?

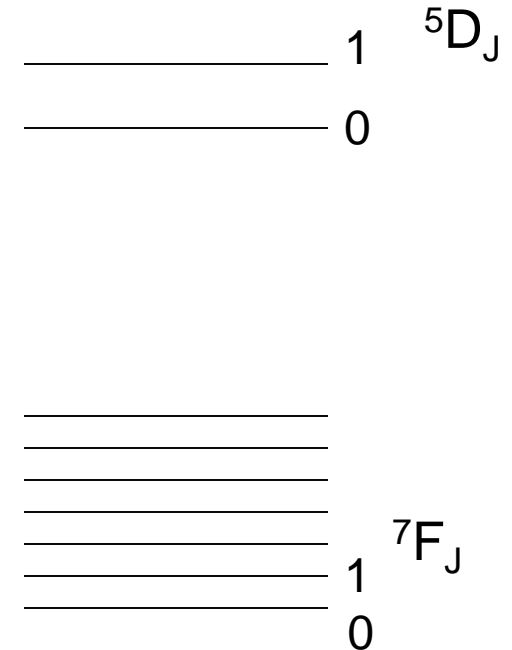


- Laser wavelength: 500nm
- Wavenumber:  $20,000\text{cm}^{-1}$
- $1\text{cm}^{-1} = 30\text{ GHz}$
- Typical optical frequencies:  
 $10^{14}\text{Hz}$
- High resolution spectroscopy:
  - MHz resolution



# Ionic energy levels

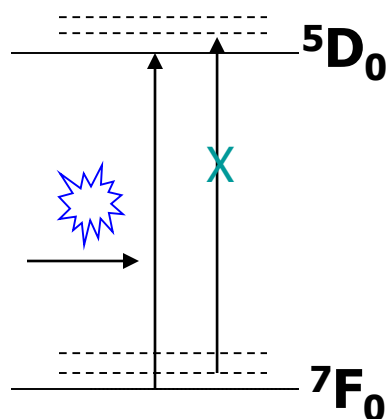
- Rare-earths: in general forms trivalent compounds
- $\text{Sm}^{2+}$  and  $\text{Eu}^{2+}$  also possible (special conditions)
- Example: Europium, at. No.=63
- Electron configuration of  $\text{Eu}^{3+}$
- $1s^2 2s^2 2p^6 3s^2 3p^6 4s^2 3d^{10} 4p^6 5s^2 4d^{10} 5p^6 4f^6$
- $\text{Eu}^{3+}$  levels (free ion)
- $^5D_J$  ( $J=0,1,2,3$ ),  $^7F_J$  ( $J=0,1,2,3,4,5,6$ ), ----
- $\text{Eu}^{3+}$  doped solids: Crystal field splitting
  - Example:  $J=1$ ; three components ( $2J+1$ )
- Hyperfine splitting





# Zero Phonon Line (ZPL)

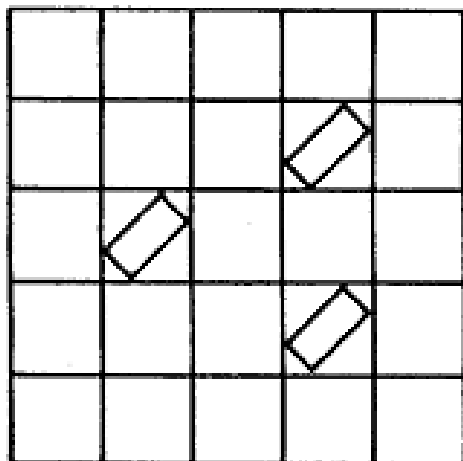
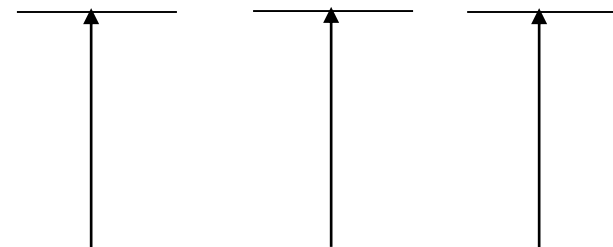
- ZPL: Pure electronic transition



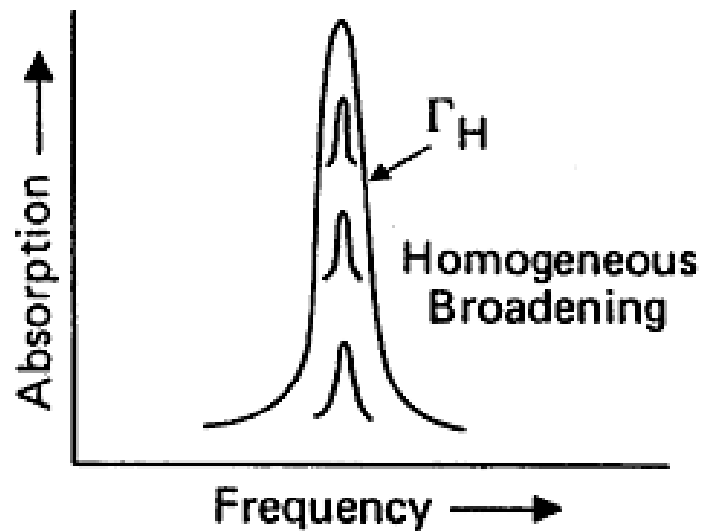
$\text{Eu}^{3+}$  ion

# Homogenous broadening

Energy levels at different centers



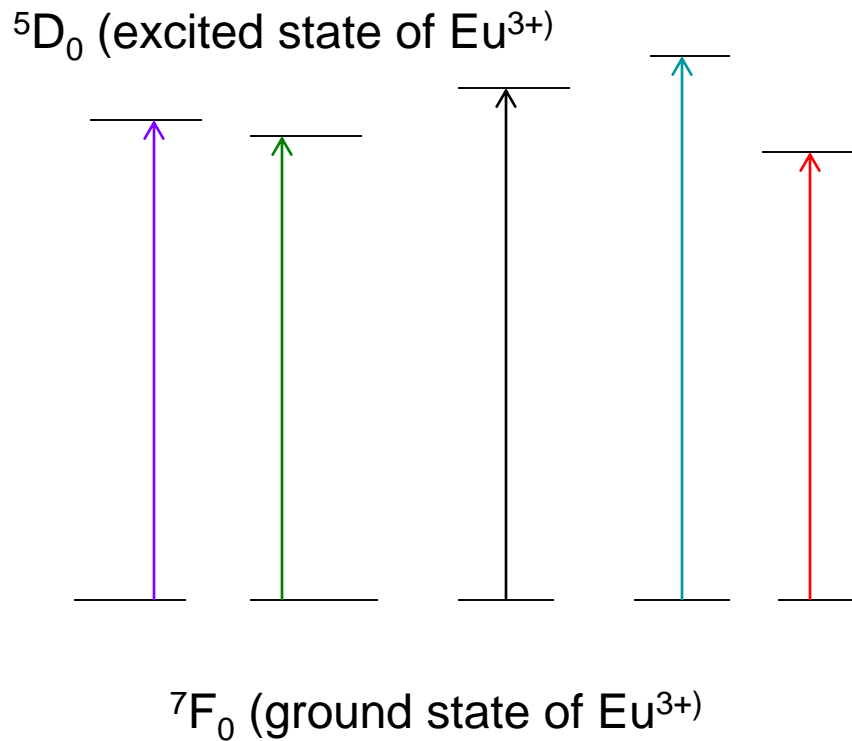
perfect crystal



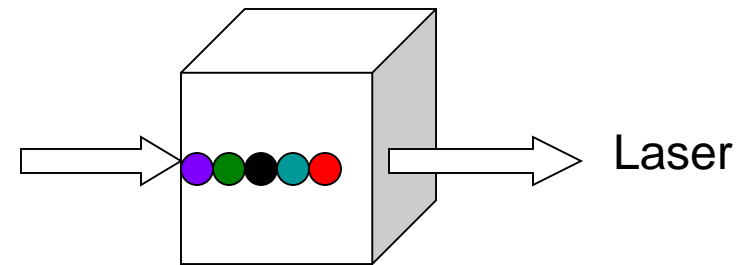


# Inhomogeneous broadening

Energy levels at different sites



Inhomogeneous crystal field

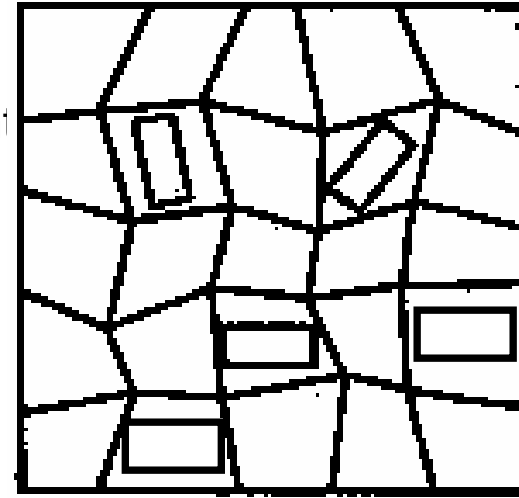
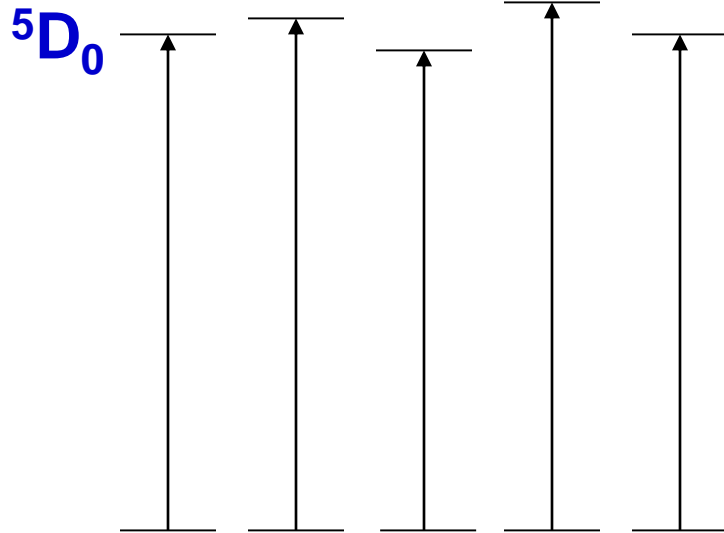


Crystal field is different at each ion  
So energy gap is different at each ion

Typical concentration:  $10^{17} - 10^{20}/\text{cc}$



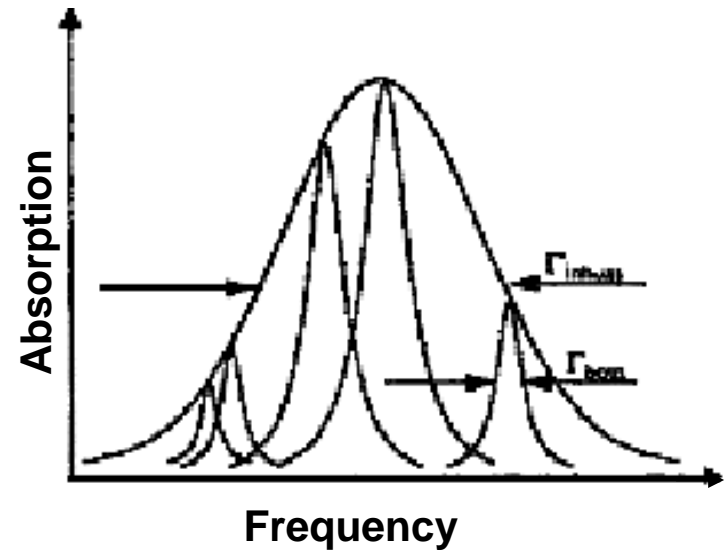
Energy levels at different



real solid matrix

${}^7F_0$  (ground state of  $\text{Eu}^{3+}$ )

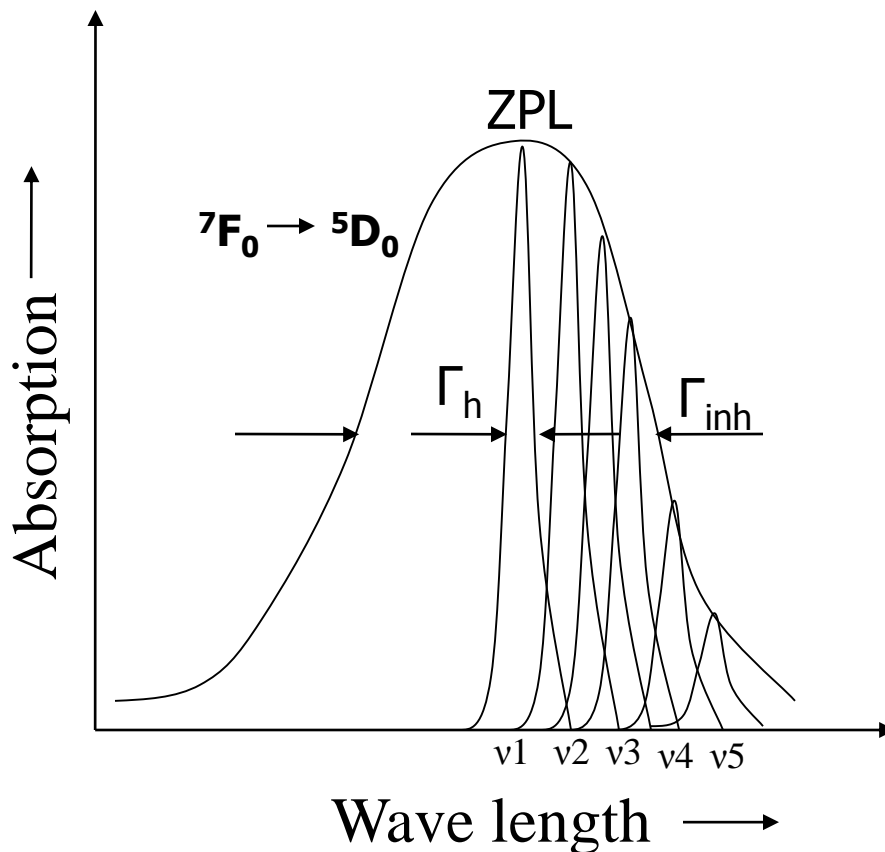
Different ions are in different crystal fields Energy gap are slightly different at each ion



**Inhomogeneous broadening**



# Optical absorption transitions



Each subset of ions absorb light at a specific frequency -  $\nu$

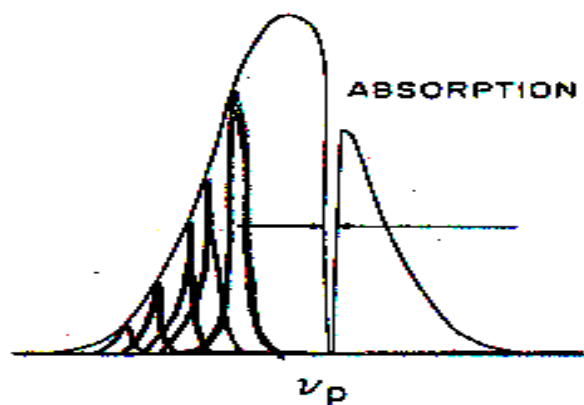
$\Gamma_{inh}$  : 1GHz – 1THz

$\Gamma_h$  : 1KHz – 100MHz

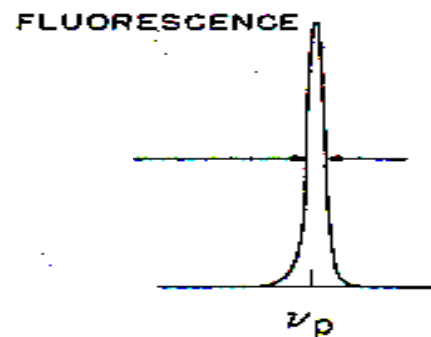
Number of Channels:  $\Gamma_{inh} / \Gamma_h$

$$\Gamma_h \propto \mathbf{1/\tau}$$

# Fluorescence line narrowing



Inhomogeneous absorption



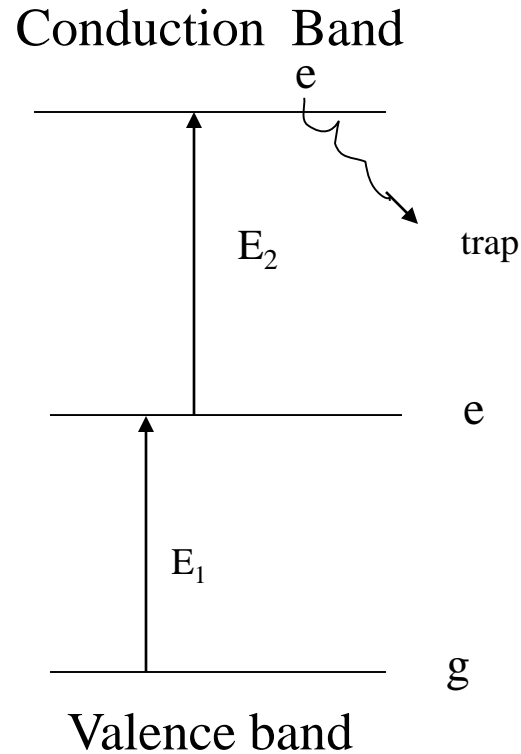
FLN



# Optical hole burning

- **High resolution** spectroscopy technique: investigate ZPL
- Application: high density optical memory
- HB: selective bleaching of a subset of ions within the ZPL (inhomog.)
- Spectral hole burning
  - Transient Hole burning (opt.pump.quadrupole levels)
  - Persistent spectral hole burning
    - » Photophysical (change in environment, TLS interaction)
    - » Photochemical(ex: photoionization, bond breakage)
    - » Gated hole burning (Ex: two-photon ionization)

# Photon gated hole burning



Winnacker et al, OL (1985); BaFCl:Sm<sup>2+</sup>  
Kaplyanskii, J.Lumin. (1998); CaS:Eu<sup>2+</sup>



# Required criteria for hole burning

- ZPL: Inhomogeneous broadening
- Homogeneous linewidth < Inhomogeneous linewidth
- Narrow laser
- Exhibit FLN: narrow holes
- Material criteria
  - No cross talk between holes
  - Holes have to be permanent
  - Burn holes with low power
  - Potential to burn/retain more holes



# Optical hole burning

- HB Materials
  - Inorganic (crystals, glasses, color centers etc.,)
  - Organics (dye doped polymers etc.,)
  - Biological
- Drawback: works at low temp.
- Our effort: make materials for high temp. HB
  - Understand the HB mechanisms
  - Improve the HB efficiency, hole density





# Why to investigate glasses?

Large inhomogeneous broadenings  
Vary the composition  
Easy to prepare samples  
PSHB

Glasses investigated:

Sodium borates  
Silicates  
Borosilicates  
Germanates  
Tellurites

# History of hole burning

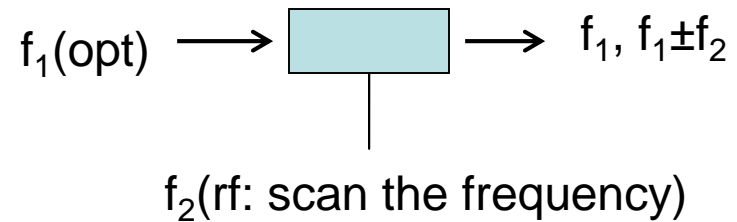
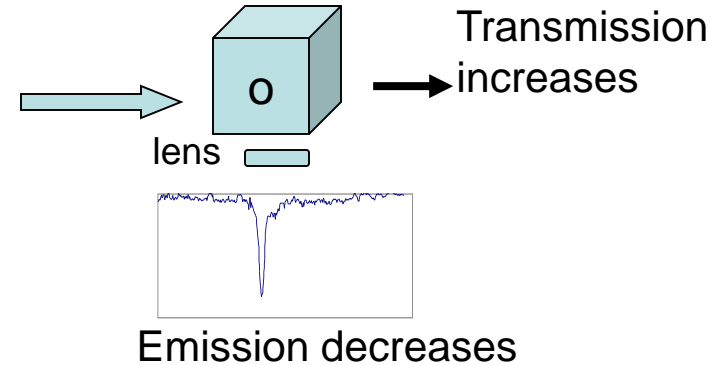
- **Pioneering HB work**
- Ruby-  $\text{Al}_2\text{O}_3:\text{Cr}^{3+}$  (A. Szabo) 1972 (patented)
- $\text{LaF}_3:\text{Pr}^{3+}$  (Erickson)
- $\text{Eu}^{3+}$
- More literature:  $\text{Pr}^{3+}$ ,  $\text{Eu}^{3+}$ : R6G dye laser
- Organics (K. Rebane group)
  
- Sm-doped glasses
  - $\text{Sm}^{2+}$ : doped borate glass (1993)
  - $\text{Sm}^{2+}$ : fluorohafnate glass (1993)
  - $\text{Sm}^{2+}$ :  $\text{SrFCl}_x\text{Br}_{(1-x)}$  (1995)

# HB in Eu-doped glasses

- Macfarlane et al, silicate (1983); <20K
- Mao et al, aluminosilicate (1996); 77K
- Fujita et al, sod. Alumin. Silicate (1998), RT, phot.red.
- Fujita et al, sodium borate (2001); 77k – RT
- Nogami et al, Sol-gel: Eu photoinduced rearrangement of protons/local structure (1998)--- rearrangement of OH bonds
- Meltzer et al, (1999); Eu<sup>3+</sup> nanoparticles (sol-gel, Y<sub>2</sub>O<sub>3</sub>)

# How to burn and probe the hole?

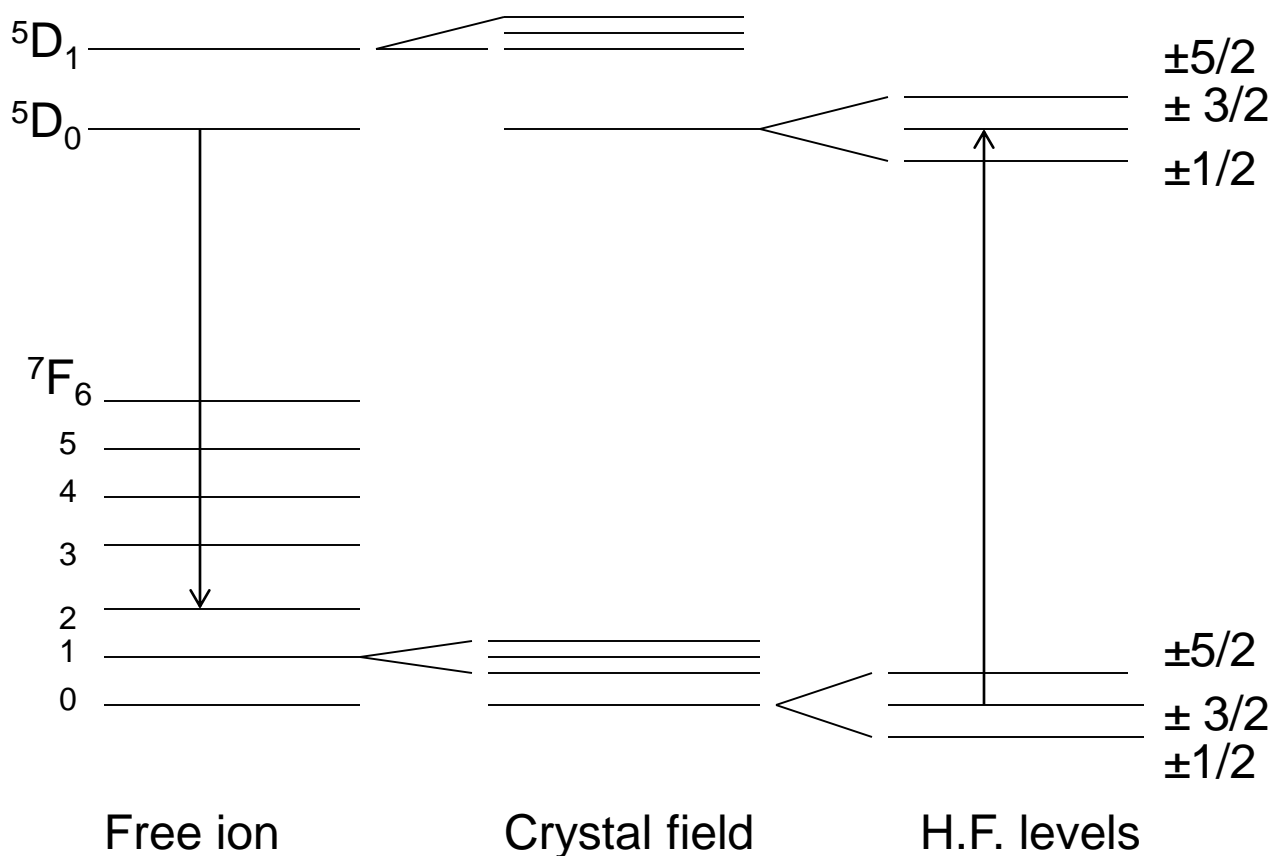
- Saturate absorption
- Excitation spectrum
  - Single laser to burn and probe
- Sideband spectroscopy
  - Use carrier frequency to burn the hole
  - Sideband frequency to probe the hole
- rf-optical double resonance





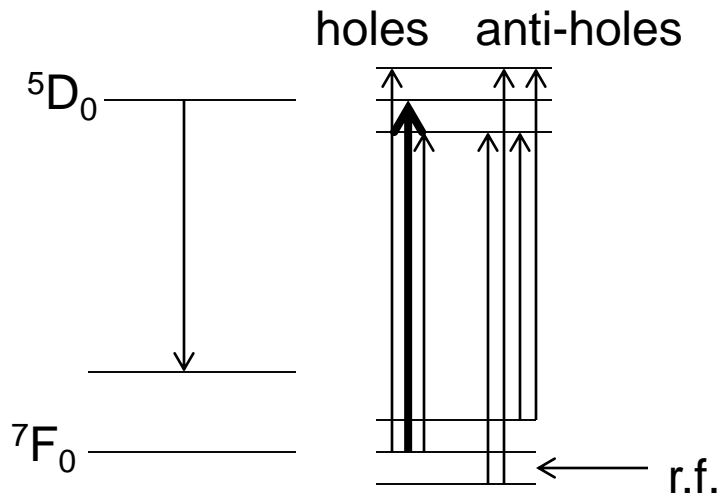
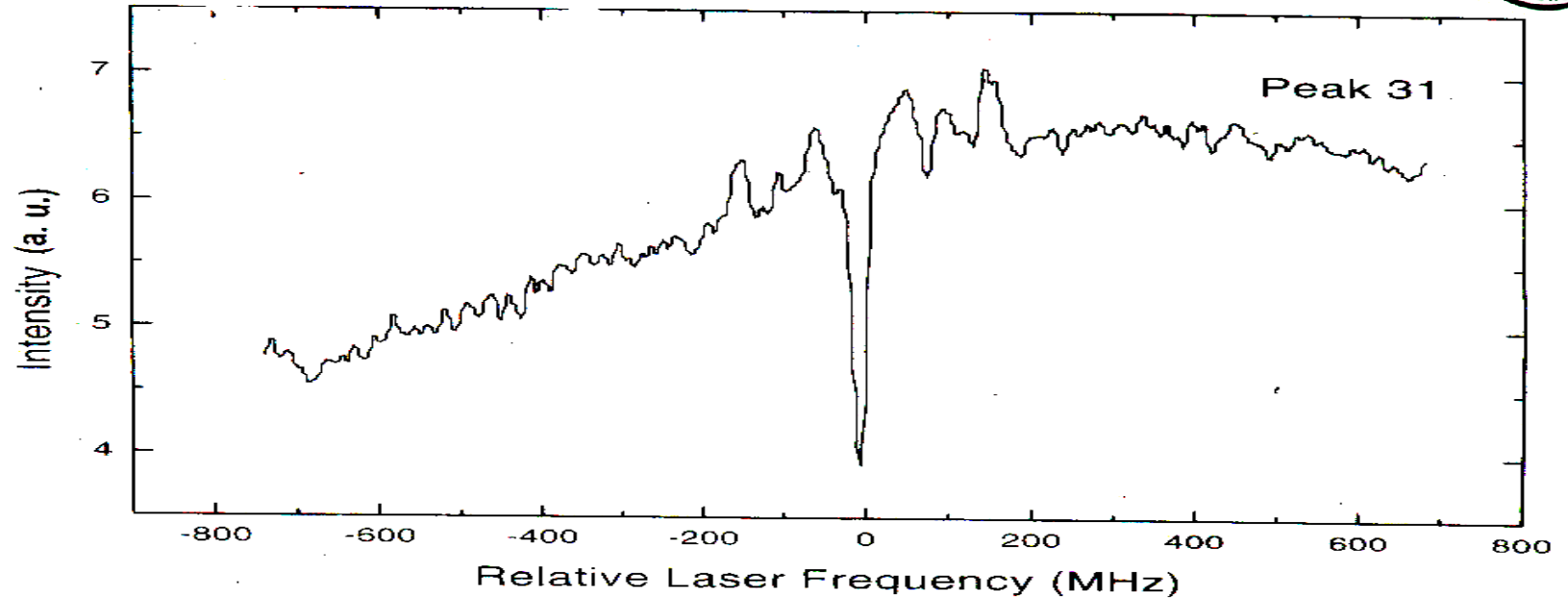
# Transient hole burning in $\text{Eu}^{3+}$

- Partial energy level diagram





# Transient HB: Population rearrangement among hf levels

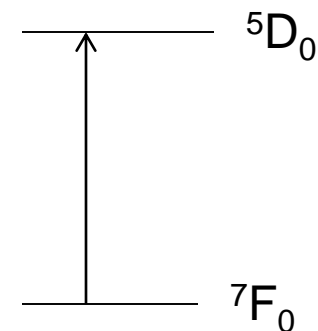
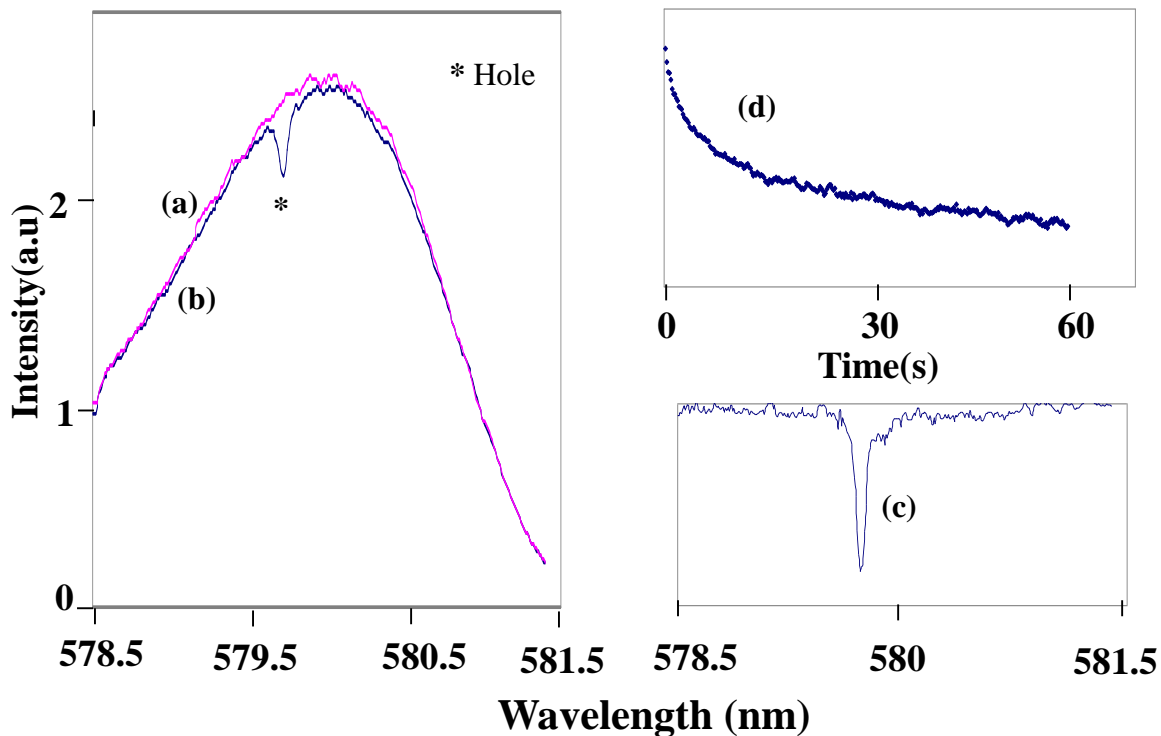


$\text{Eu}^{3+}$ -doped crystal

Appl. Opt. (1999)



# Hole burning spectra(10K)



**Excitation spectra of 612nm  $\text{Eu}^{3+}$  emission**

**(a) before and (b) after hole burning in Sodium Silicate glass**

**(c) Spectral hole**

**(d) Intensity vs Time plot obtained during hole burning**

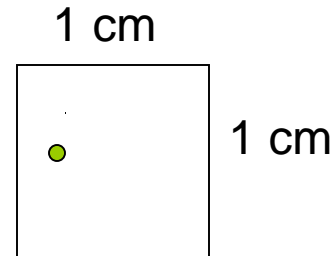


# Application: optical memory concept

$$\text{Number of holes} = \frac{\text{inhomog. width}}{\text{hole width}} = \frac{5 \times 10^9}{10^4} = 5 \times 10^5$$

Spot size =  $1 \mu\text{m}$

Spot Area =  $10^{-8} \text{cm}^2$



$$\text{Total spots} = \frac{\text{sample area}}{\text{spot area}} = \frac{1 \text{ cm}^2}{10^{-8} \text{cm}^2} = 10^8$$

Total holes = total spots  $\times$  holes at each spot

Each hole = one bit

$$\text{Total bits} = 10^8 \times 5 \times 10^5 = 5 \times 10^{13} \text{ cm}^{-2}$$

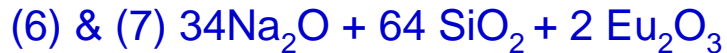


# Sample preparation: glass compositions

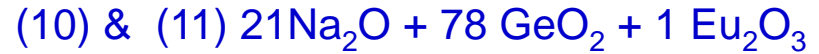
## Eu ion doped sodium borates



## Eu ion doped sodium silicates



## Eu ion doped sodium germanates



**Samples 1, 6, 8, 10, 12 were made in ambient air and**

**Samples 2, 3, 4, 5, 7, 9, 11, 13 were made in reduced ( $\text{N}_2 + \text{H}_2$ ) atmosphere**

# Material Preparation



**High temperature box and tubular furnaces**

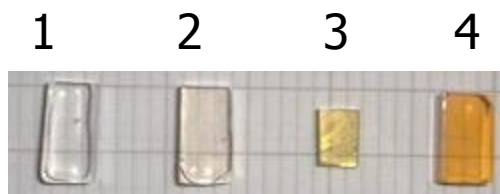


# Germanate Glass-Composition

	S#	GeO <sub>2</sub> (mol%)	Na <sub>2</sub> CO <sub>3</sub> (mol%)	Y <sub>2</sub> O <sub>3</sub> (mol%)	Eu <sub>2</sub> O <sub>3</sub> (mol%)
	1	78	21	0	1
	2	73	21	5	1
	3	78	21	0	1
	4	75	21	3	1

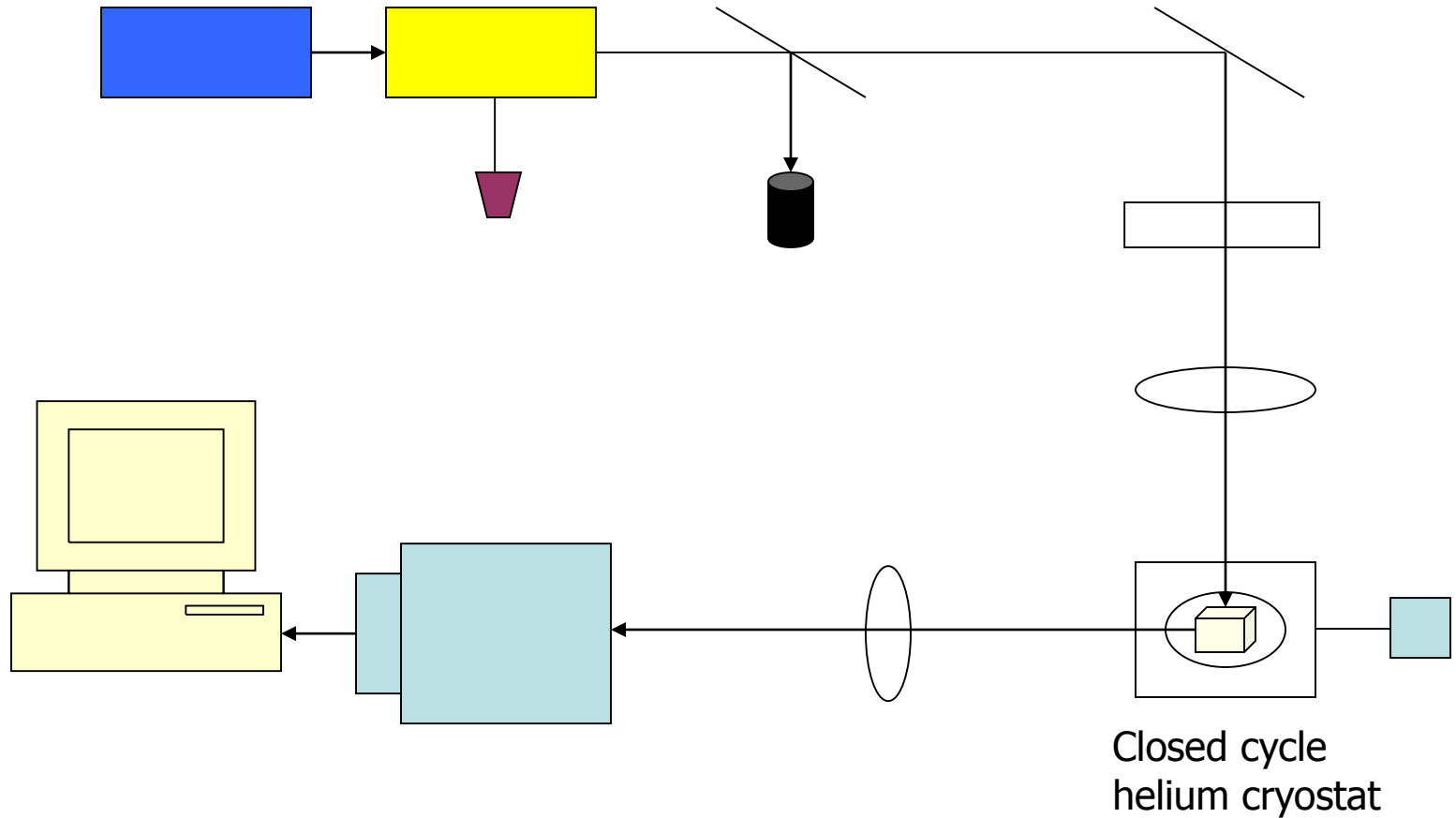
Samples 1 and 2 are melted in air

Samples 3 and 4 are prepared in reduced atmosphere

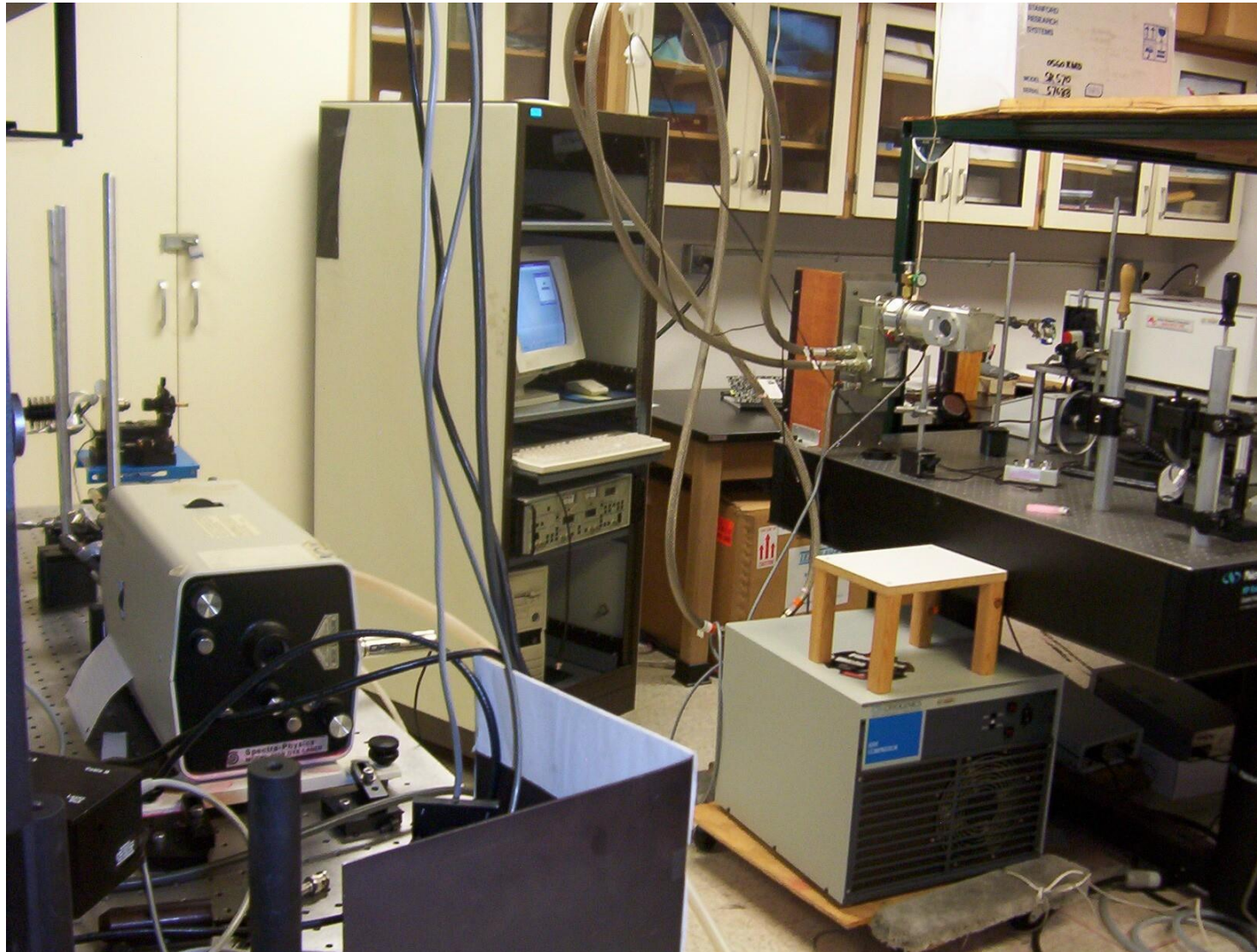




# Experimental Setup

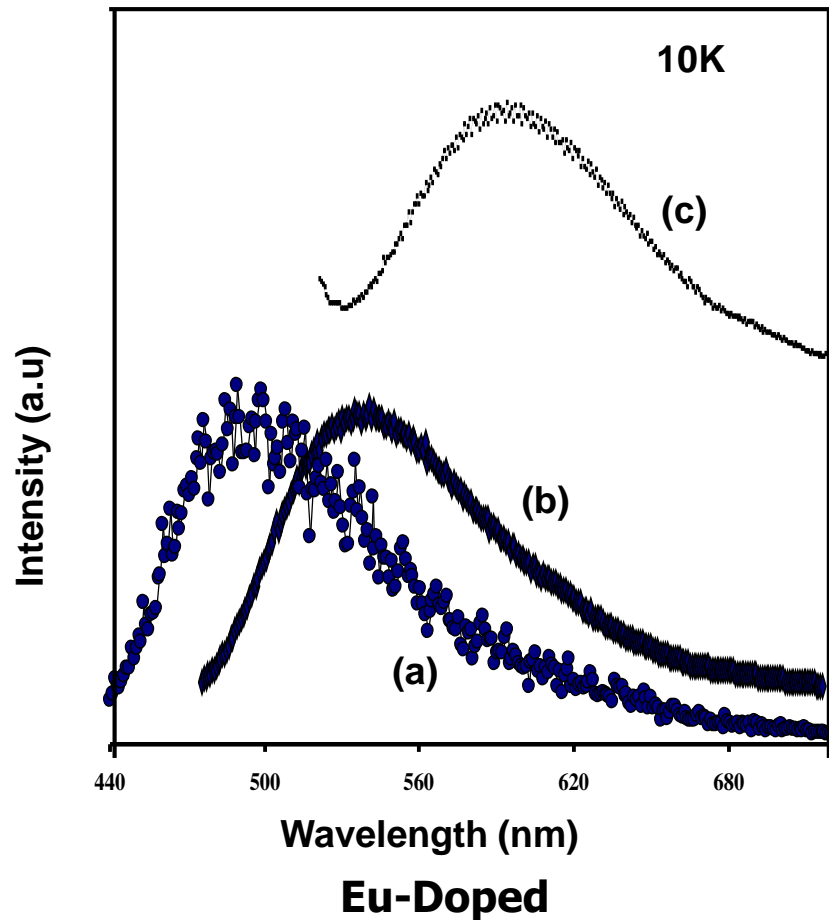


# Lab Set-Up





# Eu<sup>2+</sup> Fluorescence (4f<sup>6</sup>5d<sup>1</sup> → 4f<sup>7</sup>) in borosilicate Glass



Fluorescence Spectra of Eu<sup>3+</sup>-doped sodium borosilicate glass prepared in reduced atmosphere.

Curve (a) 337nm Laser Excitation

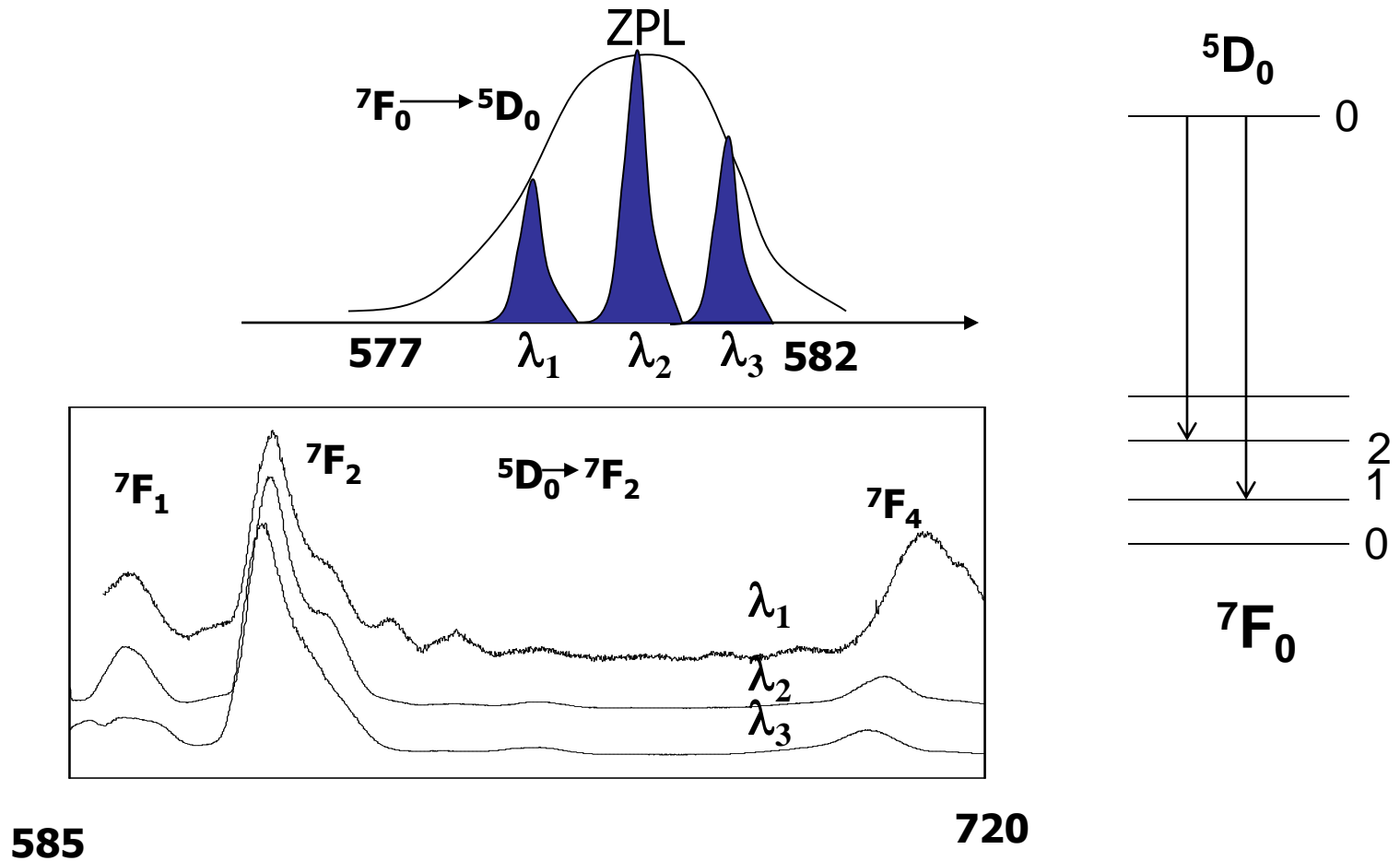
Curve (b) 458nm Laser Excitation

Curve (c) 488nm Laser Excitation

The broad absorption is due to the presence for Eu<sup>2+</sup>.



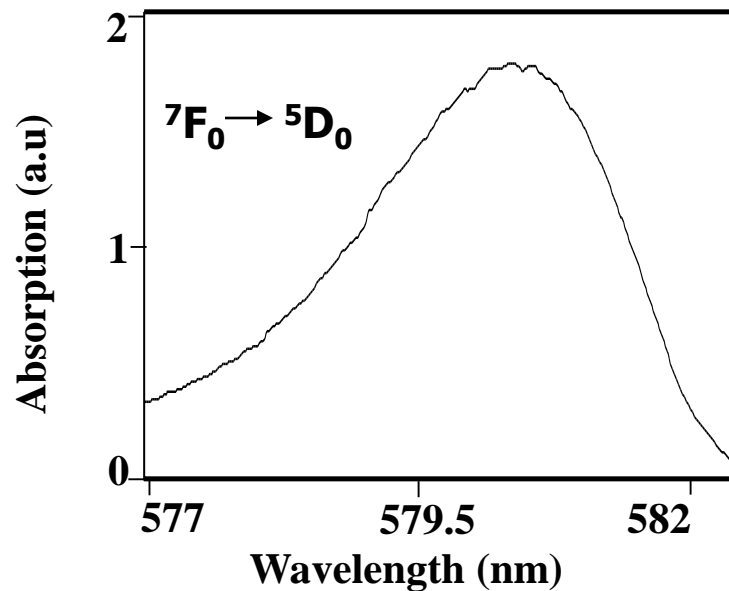
# Eu<sup>3+</sup> fluorescence ( $4f^6: {}^5D_0 \rightarrow {}^7F_J$ )



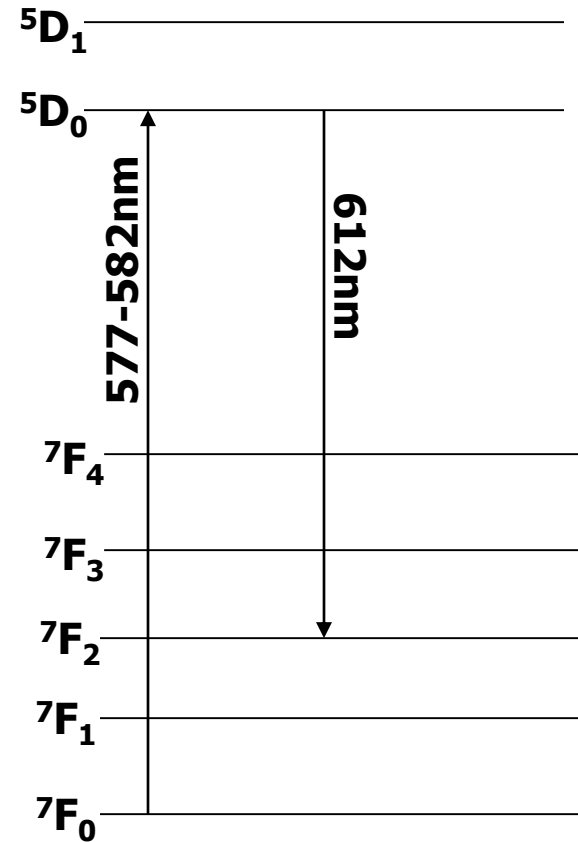
Exhibits site to site variations in the crystal field



# Excitation Spectrum

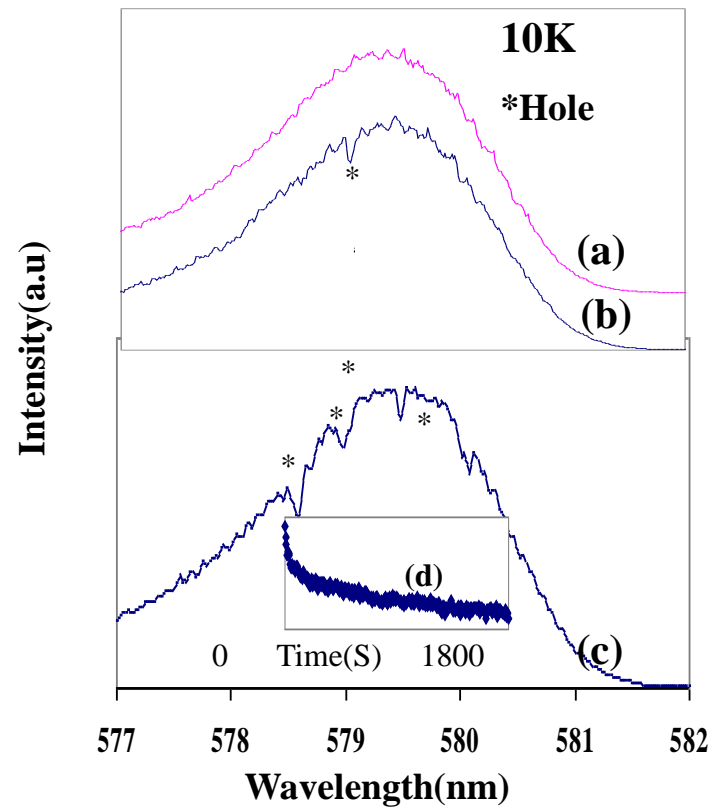


Spectrum: Recording a ZPL with a low laser power.





# HB in Eu-doped sodium borate glass (reduced)



(a) & (b) Eu

(c) Eu,Y-co-doped

## Glass composition

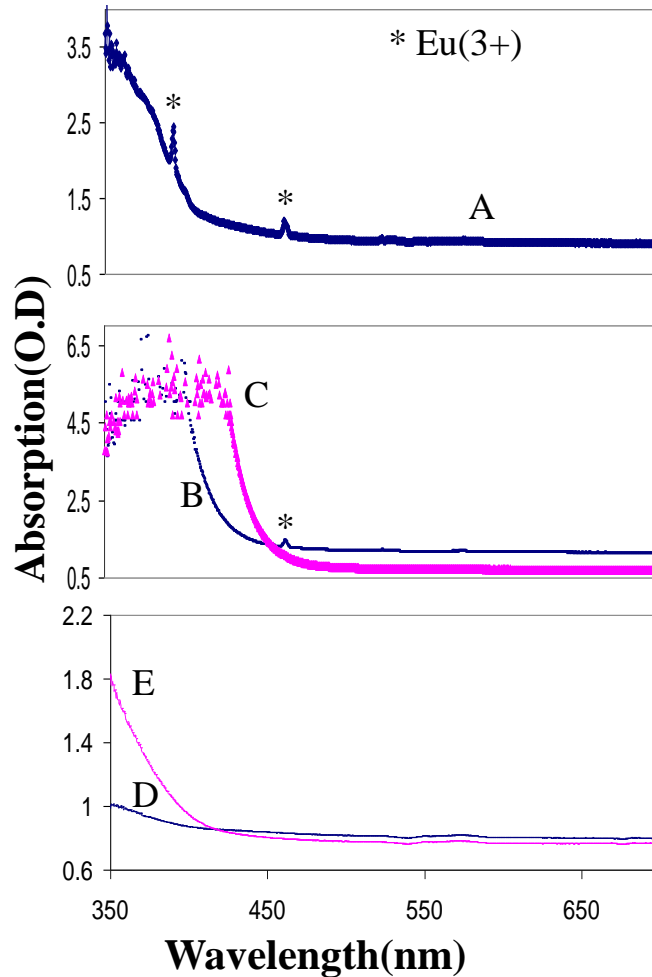
(a)  $\text{Na}_2\text{O}$ (33%),  $\text{B}_2\text{O}_3$  (65%),  $\text{Eu}_2\text{O}_3$  (2%)

(b)  $\text{Na}_2\text{O}$ (27%),  $\text{B}_2\text{O}_3$ (66%),  $\text{Y}_2\text{O}_3$ (5%),  $\text{Eu}_2\text{O}_3$ (2%)





# RT absorption spectra



(a)  $34\text{Na}_2\text{O} + 65\text{B}_2\text{O}_3 + 1\text{Eu}_2\text{O}_3$

(b)  $33\text{Na}_2\text{O} + 65\text{B}_2\text{O}_3 + 2\text{Eu}_2\text{O}_3$

(c)  $27\text{Na}_2\text{O} + 66\text{B}_2\text{O}_3 + 5\text{Y}_2\text{O}_3 + 2\text{Eu}_2\text{O}_3$

(d)  $33.6\text{Na}_2\text{O} + 66.4\text{B}_2\text{O}_3$

(e)  $27.5\text{Na}_2\text{O} + 67.4\text{B}_2\text{O}_3 + 5.1\text{Y}_2\text{O}_3$

Sample (a) was made in air atmosphere and others (b, c, d, e) were made in reduced atmosphere

J. Appl. Phys. 94, 2139 (2003)



# Hole burning efficiency

$$\eta = \frac{dT/dt}{(I/h\nu)\sigma T_0(1-T_0-R_0)}$$

$$\eta_2 / \eta_1 > 10$$

$\eta_1$  Eu doped glass

$\eta_2$  Eu, Y co-doped glass

Reason:

**Eu<sup>2+</sup> and defect concentrations: higher in the co-doped glass**

# HB mechanism: charge transfer

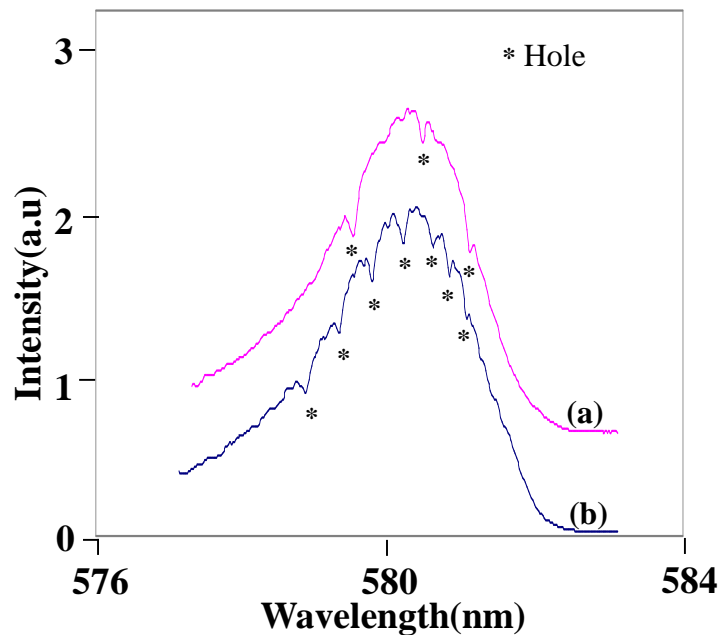
- $\text{Eu}^{3+*} + \text{defect (e}^-) \rightarrow \text{Eu}^{2+*} + \text{defect}$ 
  - $\text{Eu}^{2+}$  increases and  $\text{Eu}^{3+}$  decreases
  - Emission intensities are different before and after hole burning
- $\text{Eu}^{3+*} + \text{Eu}^{2+} \rightarrow \text{Eu}^{2+*} + \text{Eu}^{3+}$ 
  - No change in concentration
  - Emission intensity remains the same before and after hole burning
  - Agrees with the experiment



## Glass composition

(a) Na<sub>2</sub>O(34%), SiO<sub>2</sub>(64%), Eu<sub>2</sub>O<sub>3</sub>(2%)

(b) Na<sub>2</sub>O<sub>3</sub>(33%), SiO<sub>2</sub> (61%),Y<sub>2</sub>O<sub>3</sub>(4%), Eu<sub>2</sub>O<sub>3</sub>(2%)

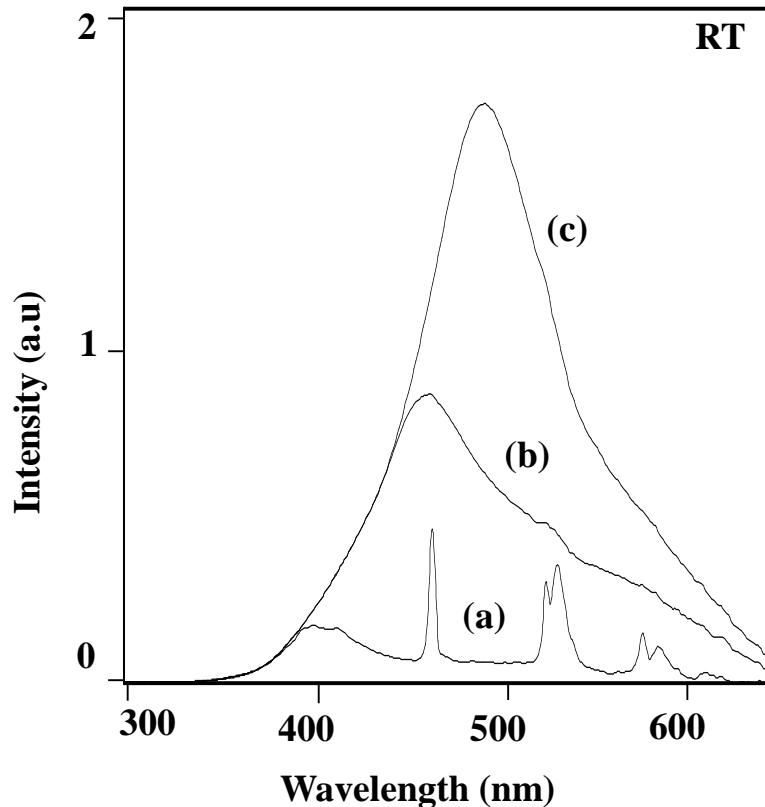


## HB in sodium silicate glass

(a) Eu (b) Eu,Y co-doped glasses

Note: HB is possible in a reduced sample

# Absorption Spectra-Borosilicate Glass



Room-temperature absorption spectra of Eu-doped glasses. Curve (a) Eu, air

Curve (b) Eu, reduced

Curve (c) Eu, Y, reduced

Sharp peaks are those of  $\text{Eu}^{3+}$  and the broad absorption is due to the presence for  $\text{Eu}^{2+}$ .

Addition of yttrium – uniform distribution of europium in the glass

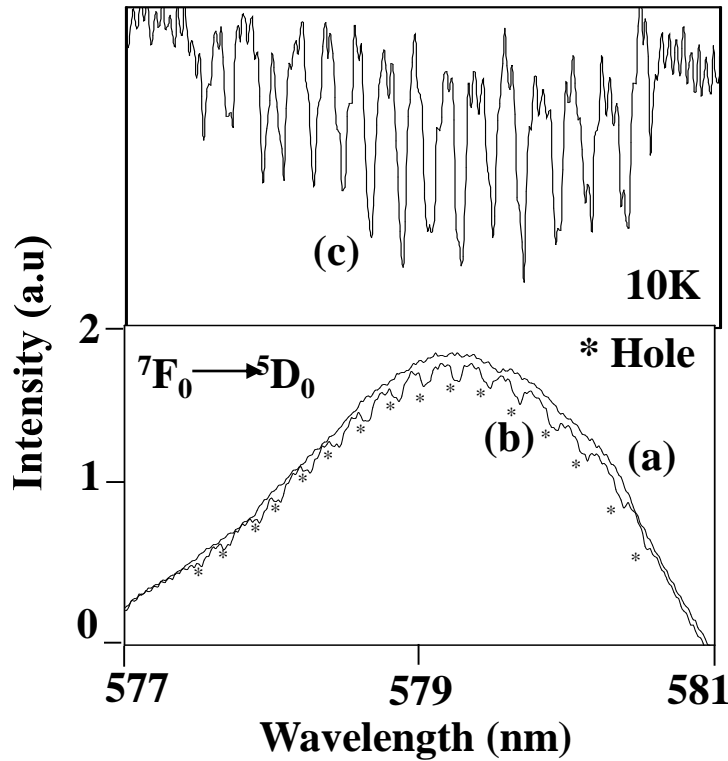


(Poster)

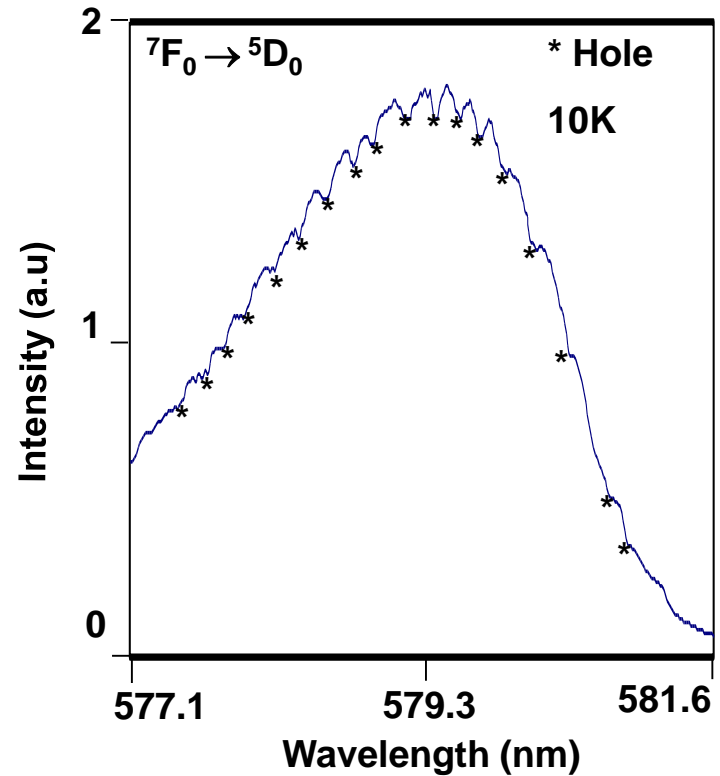
# Multiple Hole Burning in Borosilicate Glass

Glass composition

Na<sub>2</sub>O(30%), B<sub>2</sub>O<sub>3</sub>(34%), SiO<sub>2</sub>(34%), Eu<sub>2</sub>O<sub>3</sub>(2%)

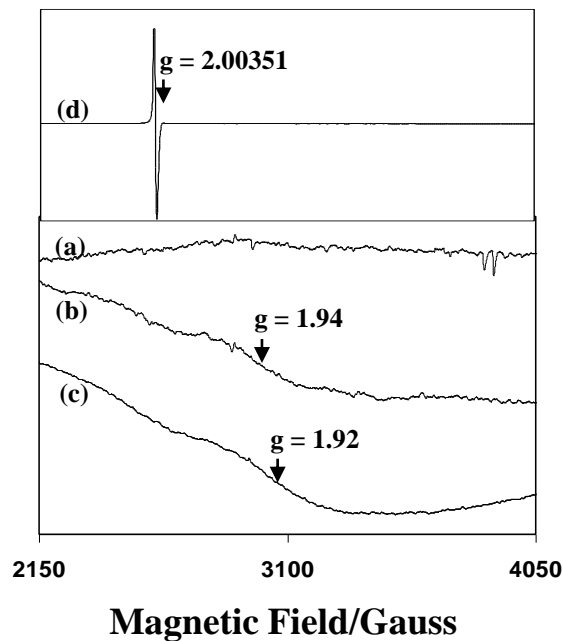


Eu<sup>3+</sup>-Doped



Eu<sup>3+</sup>, Y<sup>3+</sup> Co-doped





## ESR Spectra - Sodium silicate

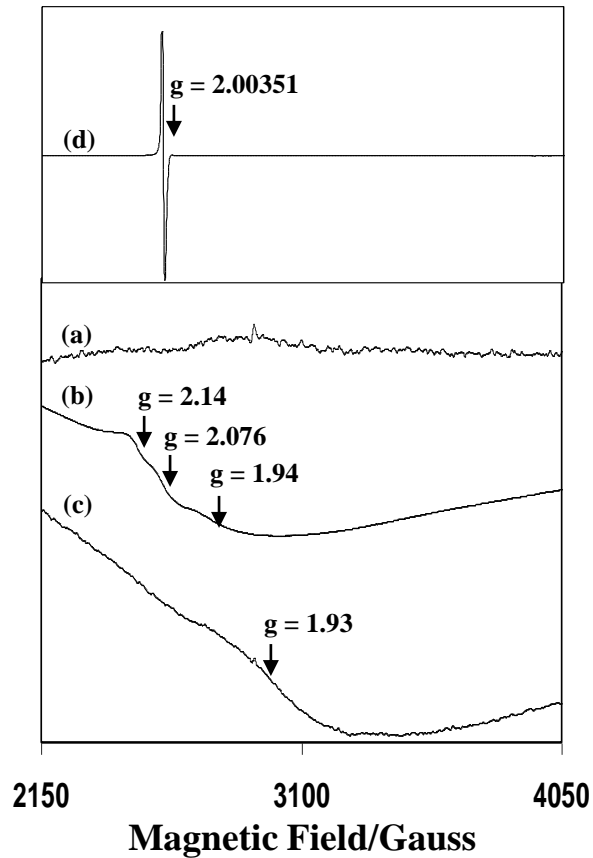
(a) Eu-doped , air

(b) Eu-doped ,  $N_2+H_2$

(c) Eu, Y co-doped,  $N_2+H_2$

(d) DPPH





## ESR—Borosilicate at 77K

(a) Eu-doped(air)

(b) Eu-doped(reduced)

(c) Eu-doped and  
Y-co-doped (reduced)

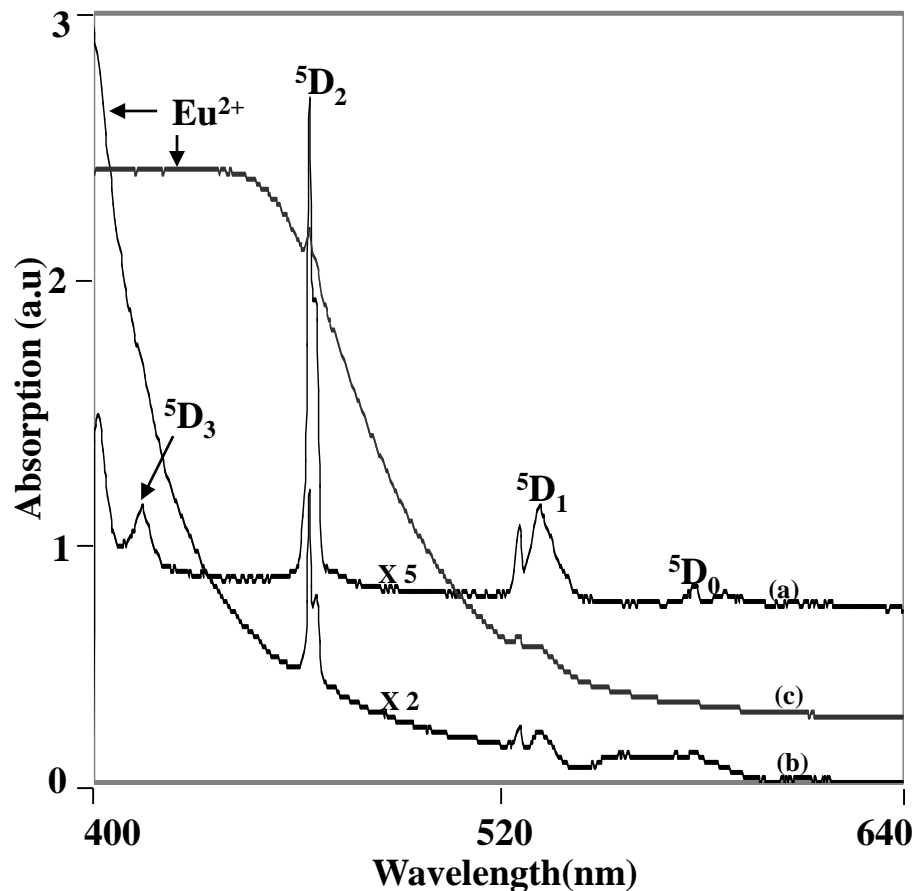
(d) DPPH

There are three types of defect centers in the borosilicate glass

It retains all the holes (higher hole retention capability)

Defects play a role in the hole burning

# Absorption Spectra-Germanate Glass



Room-temperature absorption spectra of Eu-doped glasses.  
Curve (a) Eu, air

Curve (b) Eu, reduced

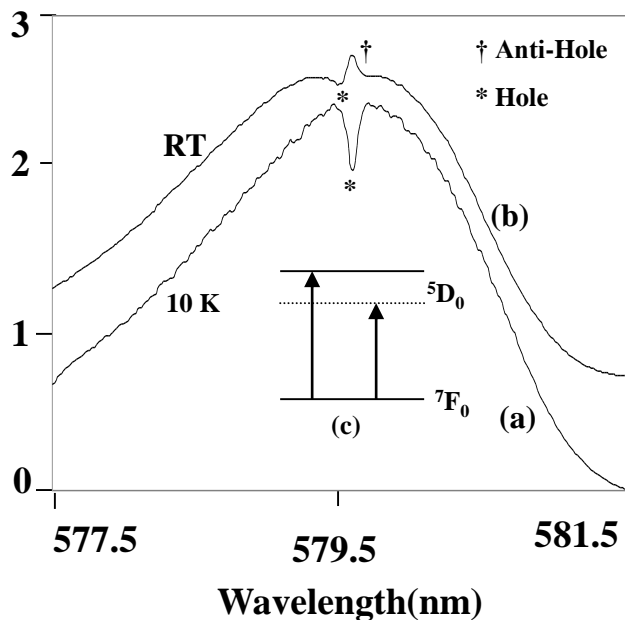
Curve (c) **Eu, Y**, reduced

Sharp peaks are those of  $Eu^{3+}$  and the broad absorption at 400nm is due to the presence for  $Eu^{2+}$ .

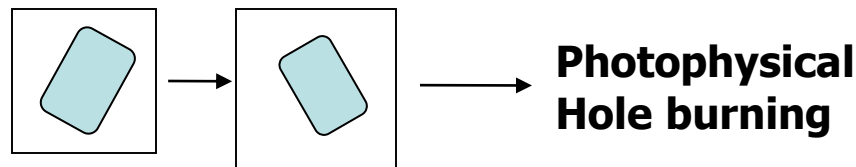
Addition of yttrium – uniform distribution of europium in the glass

# Hole Burning at 10K and 295K

Glass composition:  $\text{Na}_2\text{O}(xx\%)$ ,  $\text{GeO}_2(xx\%)$ ,  $\text{Eu}_2\text{O}_3(x\%)$



Sample	Temperature
1- Eu,air	77K
2- Eu,Y, air	77K
3- Eu, reduced	250K
4- Eu, Y, reduced	RT

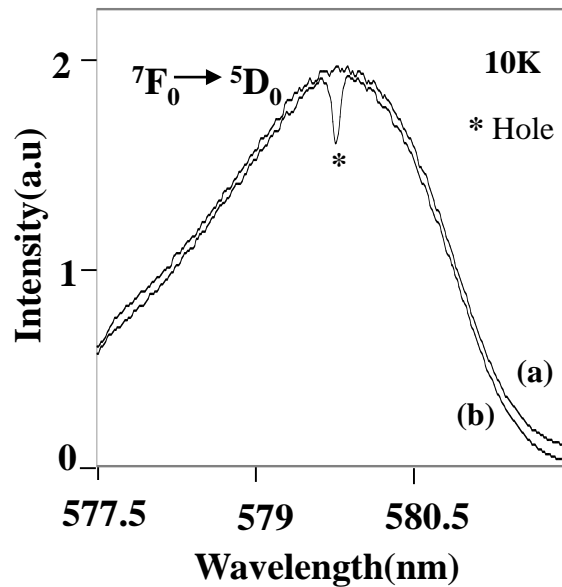


Anti-hole formation :

- (1) anti-hole is formed only when the number of ions undergoing photophysical process exceeds a certain threshold value
- (2) The photophysical process was assisted by the thermal energy

J. Appl. Phys. 96, 4012 (2004)

# Hole burning in Germanate Glass at 10K



Curve (a) excitation spectrum recorded before hole burning. Curve (b) excitation spectrum recorded after hole burning.

Sample	Dosage:	Hole depth:
1- Eu,air	50	7%
2- Eu,Y, air	1350	5%
3- Eu, reduced	25	13%
4- Eu, Y, reduced	1	18%

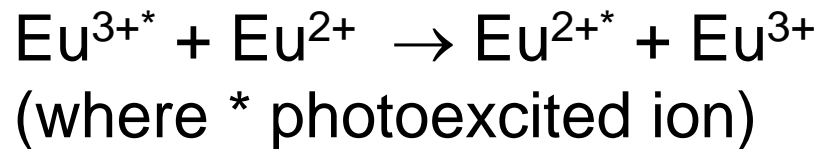
Dosage = Laser power X Time of exposure

**Mechanism : Photochemical hole burning due to a charge transfer between  $\text{Eu}^{3+}$  and  $\text{Eu}^{2+}$  ions**

**sample#1: charged defects also play a role in HB**

# Hole Burning Mechanism-Germanate Glass

- **Photophysical mechanism** – Due to change in crystal field (environmental change/or reorientation)
- **Photochemical** hole burning due to either a charge transfer between  $\text{Eu}^{3+}$  and  $\text{Eu}^{2+}$  or a defect center





# Comparison of the results-Germanate Glass

## Sample3#(Eu,Reduced)

- Min intensity : 1200 W/cm<sup>2</sup>
- Min time : 5min
- Hole width : 2.4 cm<sup>-1</sup>
- Inhomogeneous width : 64.8 cm<sup>-1</sup>
- No of holes : 10
- Hole depth : 13%
- Excitation spectrum : 250K
- I vs. T : RT

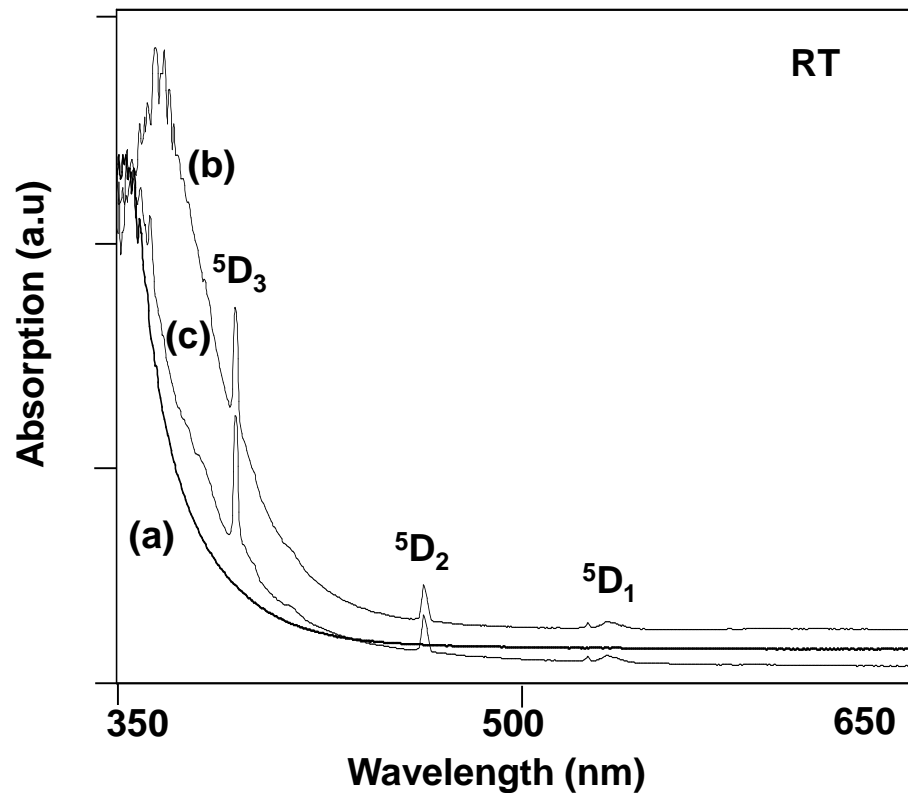
## Sample4#(Eu,Y,Reduced)

- Min intensity : 240 W/cm<sup>2</sup>
- Min time : 1min
- Hole width : 2.5 cm<sup>-1</sup>
- Inhomogeneous width : 81.6 cm<sup>-1</sup>
- No of holes : 16
- Hole depth : 18%
- Excitation spectrum : RT
- I vs. T : RT

# Absorption Spectra-Tellurite Glass



Glass composition:  $\text{Na}_2\text{O}$ (20%),  $\text{TeO}_2$ (79%),  $\text{Eu}_2\text{O}_3$ (1%)



Room-temperature absorption spectra of Undoped and Eu-doped glasses.

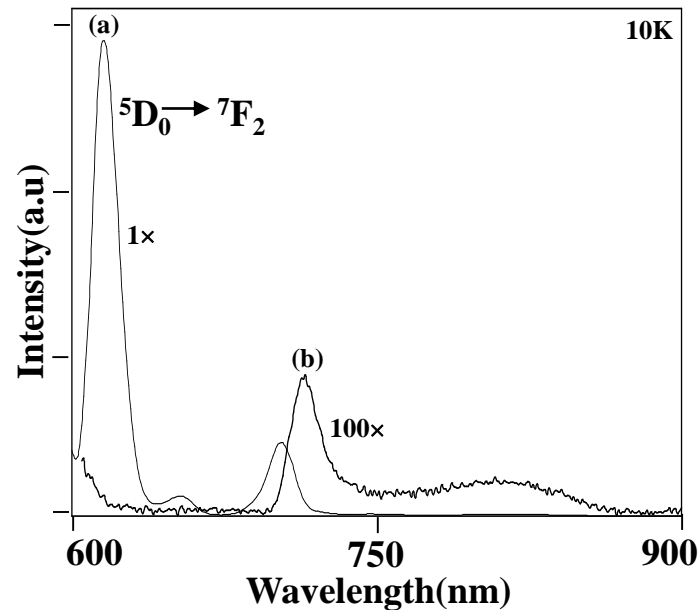
Curve (a) **No Dopant**, air – Broad absorption up to 450nm and maximum at 370nm (O.D = 4.0). This is due to defects in the host glass.

Curve (b) Eu, Y, air

Curve (c) Eu, Helium

Both reveal  $\text{Eu}^{3+}$  sharp peaks at 395, 466, 526, 535 & 580nm.

# Emission Spectra-Tellurite Glass



Fluorescence Spectra of (a)  $\text{Eu}^{3+}$ -doped and (b) undoped glasses observed under 580nm dye laser excitation.

(a)  $\text{Eu}^{3+}$  peaks at 612, 653 and 704nm.

(b) defect emission (weak) at 715, and 800nm under dye laser excitation and also at 505nm under  $\text{Ar}^+$  laser excitations



# Hole Burning in Tellurite Glass at Low and High Temperatures

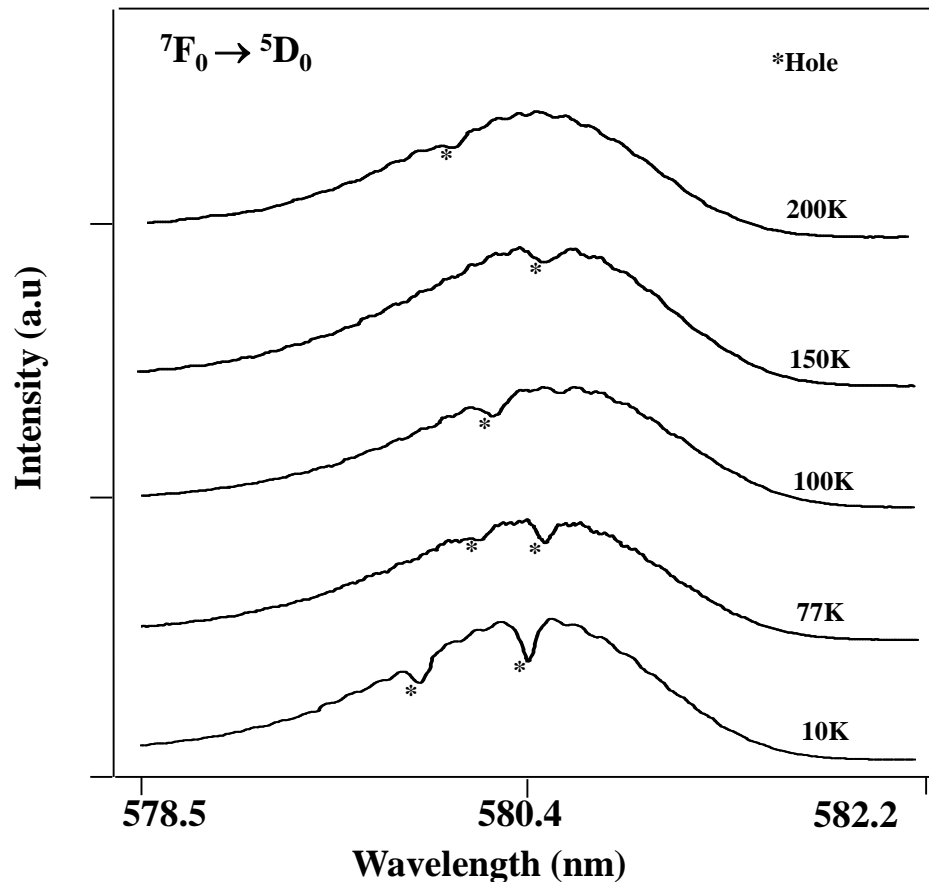


Figure shows hole burning excitation spectra recorded at various temperatures in the  $\text{Eu}^{3+}$ ,  $\text{Y}^{3+}$  co-doped sample that was melted in ambient air.

$\text{Eu}^{3+}$ -doped samples melted in air and helium atmosphere exhibited hole burning up to 100K. (Not shown here).



# Multiple Hole Burning-Tellurite Glass

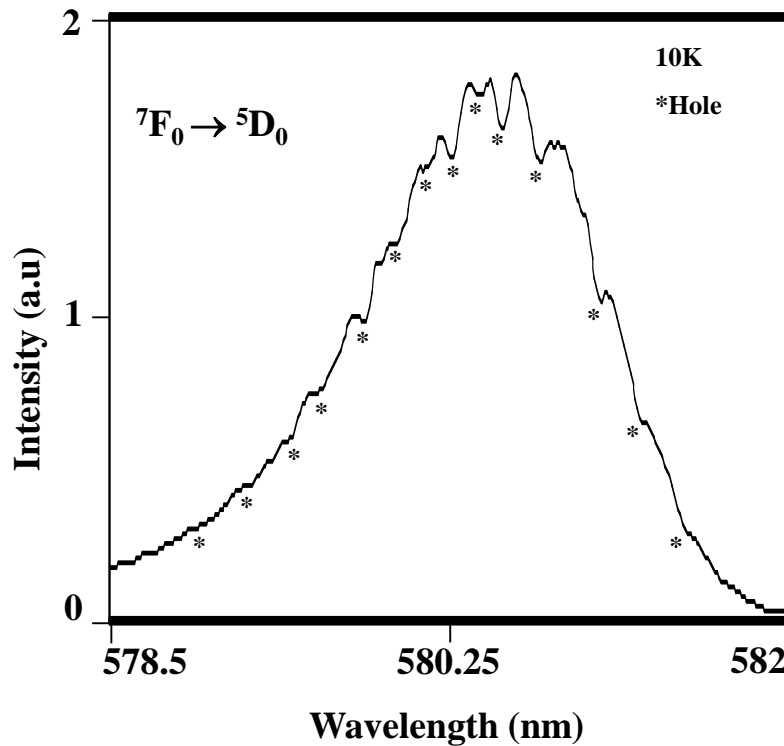
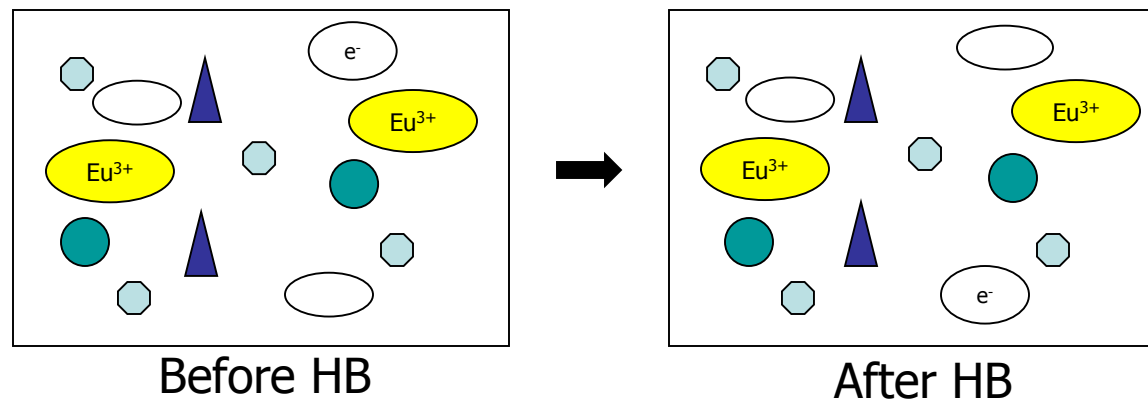


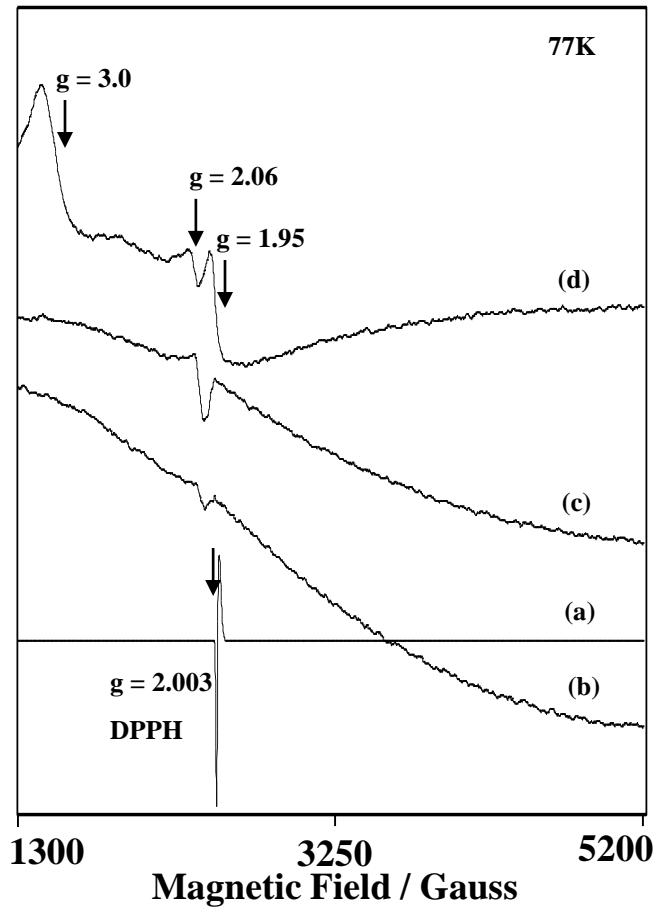
Figure shows multiple hole burning excitation spectrum recorded in the  $\text{Eu}^{3+}$ ,  $\text{Y}^{3+}$  co-doped sample that was melted in ambient air (14 holes).

$\text{Eu}^{3+}$ -doped samples melted in air and helium atmosphere also exhibited multiple hole burning at 10K. (Not shown here).

# Hole Burning Mechanism-Tellurite Glass

- PCHB: not probable because  $\text{Eu}^{2+}$  concentration is not abundant.
- No anti-hole: no evidence for a photophysical mechanism
- HB Mechanism: The motion of photoexcited defect electrons. The glass has high density of charged defects and the defect electrons migrate.





## ESR spectra of tellurite glasses

(a) Standard (b) undoped (c) Eu doped (d) Eu,Y co-doped glasses

# Comparison of the salient features of different glasses

	<u>Sample Composition</u>							$T_{\max}^a)$ (K)	$\Gamma_{\text{inh}}^b)$ ( $\text{cm}^{-1}$ )	$\Gamma_h^c)$ ( $\text{cm}^{-1}$ )	HD <sup>d)</sup> (%)	HN <sup>e)</sup>	$P_{\min}^f)$ (mw)	Atmos <sup>g)</sup>
	$\text{B}_2\text{O}_3$	$\text{SiO}_2$	$\text{GeO}_2$	$\text{TeO}_2$	$\text{Na}_2\text{O}$	$\text{Y}_2\text{O}_3$	$\text{Eu}_2\text{O}_3$							
1	65	-	-	-	33	0	2	100	62.6	1.8	11	-	100	$\text{N}_2+\text{H}_2$
2	66	-	-	-	27	5	2	300	72.0	1.8	24	-	40	$\text{N}_2+\text{H}_2$
3	-	64	-	-	34	0	2	250	62.4	2.2	27	3	1	$\text{N}_2+\text{H}_2$
4	-	61	-	-	33	4	2	295	74.6	1.9	29	7	<1	$\text{N}_2+\text{H}_2$
5	34	34	-	-	30	0	2	250	91.4	3.5	17	16	1	$\text{N}_2+\text{H}_2$
6	34	31.5	-	-	27.5	5	2	295	86.5	3.0	18	18	<1	$\text{N}_2+\text{H}_2$
7	-	-	78	-	21	0	1	295	65.4	2.3	13	10	10	$\text{N}_2+\text{H}_2$
8	-	-	75	-	21	3	1	295	80.1	2.8	18	16	5	$\text{N}_2+\text{H}_2$
9	-	-	78	-	21	0	1	77	81.1	2.9	7	11	10	Air
10	-	-	-	79	20	0	1	100	47.4	2.6	8	6	>200	Air
11	-	-	-	69	27	3	1	200	48.0	2.2	31	14	100	Air
12	-	-	-	81	18	0	1	100	47.3	2.6	9	4	>200	He



# General conclusions: Hole burning mechanisms



Sodium borate, silicate and borosilicate Glasses-

Charge transfer between  $\text{Eu}^{3+}$  and  $\text{Eu}^{2+}$  or defect

Sodium germanate Glass-

Both photophysical and photochemical processes

Sodium tellurite Glass-

Migration of photoexcited defect electrons